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Cite this article as: Rare Metal Materials and Engineering, 2019, 48(5): 1454-1460.

# First-principles Calculation of Electronic Structure and Mechanical Properties of Binary Phases in Mg-Zn-Y-La Alloy

Gao Yan<sup>1,2</sup>, Mao Pingli<sup>1</sup>, Liu Zheng<sup>1</sup>, Wang Feng<sup>1</sup>, Wang Zhi<sup>1</sup>

<sup>1</sup> Shenyang University of Technology, Shenyang 110870, China; <sup>2</sup> Shenyang Normal University, Shenyang 110034, China

**Abstract:** In order to investigate the effect of rare earth elements on the properties of Mg-Zn magnesium alloys, the structural stability, electronic structure and mechanical properties of Mg<sub>2</sub>Y, Mg<sub>2</sub>La and Mg<sub>3</sub>La compounds were calculated and analyzed by using the plane wave pseudo-potential method based on the first principles calculation. The results show that Mg<sub>3</sub>La has the strongest forming ability, Mg<sub>2</sub>La has the most stable structure by comparing the heat of formation and cohesive energy among the three compounds. The stabilization mechanism of the structure was analyzed based on the calculation of electron densities of states (DOS), electron occupation number and electron density difference. The bulk modulus *B*, shear modulus *G*, Young's modulus *E*, and Poisson's ratio v were further calculated by the elastic constants. The calculated results indicate that Mg<sub>2</sub>Y has the strongest ability of resisting deformation, Mg<sub>3</sub>La has the strongest stiffness and resistance to shear deformation, Mg<sub>2</sub>La has the strongest plasticity, Mg<sub>2</sub>Y and Mg<sub>2</sub>La are ductile phases while Mg<sub>3</sub>La is a brittle phase. In addition, the calculated results of hardness and melting temperature show that Mg<sub>3</sub>La has the largest hardness and Mg<sub>2</sub>Y has the highest melting temperature among the three compounds.

Key words: magnesium alloy; first principles calculation; structural stability; electronic structure; elastic properties

Magnesium alloy is the lightest metal material in industrial application and it has been intensively used in the automobile, aerospace and 3C industries due to its advantages of high specific strength and high rigidity, good shock resistance and machinability<sup>[1-3]</sup>. However, the inferior mechanical properties of magnesium alloys at high temperature restricted its further development. Research results show that an effective way to improve the mechanical properties of magnesium alloys was alloying magnesium alloys using rear earth elements<sup>[4]</sup>. Mg-Zn-Y ternary alloy has attracted much attention due to its excellent mechanical properties and unique crystal structure<sup>[5]</sup>. The addition of Y to the Mg-Zn alloy can not only improve the ambient mechanical properties by improving the casting properties and the formation of the stable quasicrystal strengthening phase but also improve the high temperature strength and creep resistance of magnesium alloys<sup>[6, 7]</sup>. However, the im-

provement of the properties of the alloys was limited by only one type of rear earth element addition. The research results show that the addition of multiple rare earth elements could significantly improve the properties of the magnesium alloys through the mutual influence of several elements<sup>[8]</sup>. Many researches has focused on the high cost heavy rear earth elements, such as Gd and Dy, while only little attention has been paid to low cost light rare earth elements, such as La and Ce. The structural stability and electronic structure of Mg-La intermetallic compounds were investigated<sup>[9,10]</sup>. It was found that Mg<sub>3</sub>La had the strongest forming ability and MgLa had the most stable structure. The elastic properties and electronic structure of Mg<sub>2</sub>Y and Mg<sub>2</sub>La compounds were investigated by Chen et al<sup>[11]</sup>. It was found that Mg<sub>2</sub>Y was a ductile phase, while Mg<sub>2</sub>La was a brittle phase. The calculated results of elastic and thermodynamic properties of Mg<sub>3</sub>La phase showed that

Received date: June 25, 2018

Foundation item: National Natural Science Foundation of China (51571145); Shenyang Science and Technology Plan (17-9-6-00)

Corresponding author: Mao Pingli, Ph. D., Professor, School of Materials Science and Engineering, Shenyang University of Technology, Shenyang 110870, P. R. China, Tel: 0086-24-25497131, E-mail: maopl@sut.edu.cn

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it was a brittle phase which is dynamically stable<sup>[12]</sup>. Irrespective of the above mentioned theoretical investigation on the Mg<sub>2</sub>Y, Mg<sub>2</sub>La and Mg<sub>3</sub>La compounds, no systematic theoretical investigation has been reported on structural stability, electronic structure and mechanical properties of Mg<sub>2</sub>Y, Mg<sub>2</sub>La and Mg<sub>3</sub>La laves phases in La alloying Mg-Zn-Y alloys by first principles calculations.

In order to study the stability, electronic structure and mechanical properties of  $Mg_2Y$ ,  $Mg_2La$  and  $Mg_3La$  compounds, the heat of formation, cohesive energy, density of state, Mulliken electron occupation numbers, electron density difference, elastic constants, hardness and melting temperature of  $Mg_2Y$ ,  $Mg_2La$  and  $Mg_3La$  were calculated and analyzed by using the first-principle method in the present work. The calculated results have good consistency with other theoretical and experimental values of relevant documents. We hope that the research results can serve as a theoretical guidance of the development of  $Mg_2Tn-Y$  series alloys.

# **1** Computational Methods

This study was carried out by the first principle method, which is based on density functional theory  $(DFT)^{[13]}$ , using the Cambridge Serial Total Energy Package (CASTEP) code. Through the plane wave pseudo-potentials method<sup>[14]</sup>, the exchange-correlation terms in the electron-electron interaction was treated with the Perdew-Bruke-Ernzerhof (PBE)<sup>[15]</sup> version of generalized gradient approximation (GGA). The cut-off energy of plane wave was set to 380 eV, in order to keep the total energy and function on the atomic force convergence. The total energy convergence value is  $5.0 \times 10^{-8}$  eV/atom, and the force on all atoms is less than 0.001 eV/nm. The k-points of Monkhost-Pack scheme is  $6 \times 6 \times 6$ .

# 2 Results and Discussion

# 2.1 Crystal structure and stability

The crystal structures of  $Mg_2Y$ ,  $Mg_2La$  and  $Mg_3La$  are shown in Fig.1. The structure parameters are listed in Table 1. The calculated crystal constants of  $Mg_2Y$ ,  $Mg_2La$  and  $Mg_3La$  are listed in Table 2. For the reference the calculated results by other authors using the same calculation method are also shown in Table 2. It can be seen that the present optimized results are in good agreement with other computational results, suggesting that the calculated results in this work are highly reliable. In order to compare the forming and thermodynamic stability, the heats of formation and cohesive energies of  $Mg_2Y$ ,  $Mg_2La$  and  $Mg_3La$  are calculated in the present paper by the following equations:

$$\Delta H = \frac{E_{\text{tot}}^{\text{AB}} - N_{\text{A}} E_{\text{solid}}^{\text{A}} - N_{\text{B}} E_{\text{solid}}^{\text{B}}}{N_{\text{A}} + N_{\text{B}}}$$
(1)

$$E_{\rm coh} = \frac{E_{\rm tot}^{\rm AB} - N_{\rm A} E_{\rm atom}^{\rm A} - N_{\rm B} E_{\rm atom}^{\rm B}}{N_{\rm A} + N_{\rm B}}$$
(2)

where  $\Delta H$  and  $E_{\rm coh}$  are the heats of formation and cohesive energies of intermetallic compound, respectively,  $E_{\rm tot}^{AB}$  is the total energy of alloy,  $E_{\rm solid}^{A}$  and  $E_{\rm solid}^{B}$  are the energy per atom of pure elements A and B, respectively.  $E_{\rm atom}^{A}$  and  $E_{\rm atom}^{B}$  are the energies of A and B atoms in the free state, respectively,  $N_{\rm A}$  and  $N_{\rm B}$  are the atom numbers in the unit cell.

The heats of formation and cohesive energies of three kinds of crystal structure are shown in Table 3. The easiness of formation of intermetallic compounds can be expressed by heats of formation. The lower the heats of formation are, the better the forming ability are. The negative values of the heats of formation of the three crystalline structures imply that all the three compounds can be formed by exothermic reaction in Mg-Zn-Y-La alloys. By comparison it can be seen that Mg<sub>3</sub>La is the easiest to form, followed by Mg<sub>2</sub>La and Mg<sub>2</sub>Y.

The stability of the crystal structure is decided by the cohesive energy. The definition of cohesive energy is that if the crystal is split into a single atom, the lower the cohesive energy is, the more stable the crystal structure is<sup>[19]</sup>. It is shown that Mg<sub>2</sub>La has the strongest alloying ability, followed by Mg<sub>2</sub>Y and Mg<sub>3</sub>La. For the other important structure of MgZn<sub>2</sub> in Mg-Zn-Y-La alloy, the cohesive energy has been already calculated by our research group<sup>[3]</sup>, and the value is -134.22 kJ/mol. The cohesive energies of Mg<sub>2</sub>Y, Mg<sub>2</sub>La and Mg<sub>3</sub>La are found to be much higher than that of MgZn<sub>2</sub> by comparison. Therefore, the addition of Y and La elements play a very important role in improving the stability of Mg-Zn alloy structure.

### 2.2 Electronic structures

The stability and mechanical properties of intermetallic compounds are determined by their electronic structures <sup>[23]</sup>. In order to further understand the bonding characteristics,

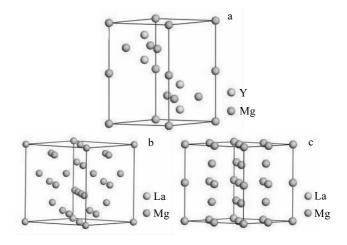


Fig.1 Crystal structures of Mg<sub>2</sub>Y (a), Mg<sub>2</sub>La (b), and Mg<sub>3</sub>La (c)

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Table 1 Structure parameters of Mg2 Y, Mg2La and Mg3La								
Phase	Atom number in cell	Space group	Structure type	Pearson sign	Atomic coordinate			
$Mg_2Y$	12	P63/MMC	C14	hP12	Mg: (0,0,0), Mg: (0.84,0.68,0.25) Y: (0.33,0.67,0.62)			
$Mg_2La$	24	FD-3M	C15	cF24	Mg: (0.625, 0.625, 0.625), La: (0,0,0)			
Mg <sub>3</sub> La	16	FM-3M	C15	cF16	Mg: (0.5,0.5,0.5) Mg: (0.25,0.25,0.25), La: (0,0,0)			

able 1 Structure parameters of Mg<sub>2</sub>Y, Mg<sub>2</sub>La and Mg<sub>3</sub>La

Table 2Crystal constant of Mg2Y, Mg2La and Mg3La

Phase		$a_0 \times 0.1 \text{ nm}$		$c_0 \times 0.1 \text{ nm}$	$V_0 / \times 10^{-3}$	- (~ ~ ~ -3	
Fliase	Present Ref.		Present Ref.		nm <sup>3</sup>	$\rho/\text{g·cm}^{-3}$	
Mg <sub>2</sub> Y	6.032	6.037 [16] , 6.064 [17]	9.960	9.752 <sup>[16]</sup> , 9.786 <sup>[17]</sup>	317.393	2.968	
Mg <sub>2</sub> La	8.827	8.769 <sup>[9]</sup> , 8.817 <sup>[17]</sup>	8.827	8.769 [9], 8.817 [17]	687.671	3.622	
Mg <sub>3</sub> La	7.537	7.495 <sup>[9]</sup> , 7.507 <sup>[18]</sup>	7.537	7.495 <sup>[9]</sup> , 7.507 <sup>[18]</sup>	428.165	3.286	

Table 3	Calculated heat of formation	$(\Delta H)$ and cohesive energy	$(E_{\text{coh}})$ of Mg <sub>2</sub> Y, Mg <sub>2</sub> La and Mg <sub>3</sub> La
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Phase		$\Delta H/kJ \cdot mol^{-1}$	$E_{\rm coh}/{\rm kJ}\cdot{\rm mol}^{-1}$			
Thase	Present	Exp.	Cal.	Present	Exp.	Cal.
$Mg_2Y$	-10.74	-10.37 [17]	-11.99 <sup>[17]</sup> , -9.17 <sup>[20]</sup>	-243.69	-	-
Mg <sub>2</sub> La	-11.77	-12.30 [21]	-11.66 <sup>[21]</sup> , -12.55 <sup>[22]</sup>	-251.78	-	-253.11 [9]
Mg <sub>3</sub> La	-12.49	-13.50, -13.30 [18]	-13.36 <sup>[18]</sup> , -9.00 <sup>[9]</sup>	-225.82	-	-229.66 [9]

the electronic structures of Mg<sub>2</sub>Y, Mg<sub>2</sub>La and Mg<sub>3</sub>La compounds were calculated and analyzed. The total and partial density of states are shown in Fig.2, and the Fermi level  $E_{\rm F}$ are indicated by the dotted line in the figures. It can be seen from Fig.2 that the peak density of states near the Fermi level for Mg<sub>2</sub>Y, Mg<sub>2</sub>La and Mg<sub>3</sub>La are all concentrated between -7 and 0 eV, implying that, Mg<sub>2</sub>Y, Mg<sub>2</sub>La and Mg<sub>3</sub>La all exhibit metallic properties. The contribution of Mg<sub>2</sub>Y bonding electrons mainly come from the orbital electrons of Mg(3s), Mg(2p), Y(5s), Y(4p) and Y(4d). The binding electrons of Mg<sub>2</sub>La and Mg<sub>3</sub>La are mainly originated from Mg(3s), Mg(2p), La(6s), La(5p), La(5d) orbital. The total density of states at the Fermi level are also shown in Fig.2. It is found that the states density of the Mg<sub>2</sub>Y, Mg<sub>2</sub>La and Mg<sub>3</sub>La compounds at the Fermi level are 13.70, 24.678 and 8.96, respectively. From this, it can be concluded that Mg<sub>2</sub>La has the highest conductivity in the three compounds.

The stability of the crystal structure is also closely related to the number of bonding electrons. The more electrons are bonded, the stronger the interaction between the charges and the more stable of the structure is<sup>[24]</sup>. The number of bonded electrons between -7 and 0 eV for Mg<sub>2</sub>Y is 75.98, Mg<sub>2</sub>La is 215.92 and Mg<sub>3</sub>La is 35.09. It can be seen from the interaction between the charge available that Mg<sub>2</sub>La has the highest stability, followed by Mg<sub>2</sub>Y, and then Mg<sub>3</sub>La. Considering the combination of cohesive energy and electronic structure, it can be deduced that Mg<sub>2</sub>La has the highest stability among the three compounds.

The bonding strength between atoms can be analyzed by electron occupation number<sup>[25]</sup>. The calculated results of the

electron occupation numbers of Mg<sub>2</sub>Y, Mg<sub>2</sub>La and Mg<sub>3</sub>La are shown in Table 4. From the table it can be seen that for Mg<sub>2</sub>Y, the electrons transfer not only takes place from Y to Mg atoms, but also between Mg and Mg atoms. As for Mg<sub>2</sub>La electrons transfer from La to Mg atoms. Whereas for Mg<sub>3</sub>La electrons transfer from Mg to La atoms. Considering the atoms number, the average charge transfer between different types of atoms in the three compounds is 0.043 for Mg<sub>2</sub>Y, 0.19 for Mg<sub>2</sub>La, and 0.047 for Mg<sub>3</sub>La. In the three kinds of crystal structures, the order of the ionic bond is Mg<sub>2</sub>La>Mg<sub>3</sub>La >Mg<sub>2</sub>Y, which is in agreement with the results of bonding electron number analysis. Based on the above analysis, it can be found that although Mg<sub>3</sub>La has the strongest formation ability, the stability of Mg<sub>3</sub>La is weaker than that of Mg<sub>2</sub>La.

The bonding characteristics of the intermetallic compounds can also be visualized by the electron density difference. In this paper, the electron density difference of Mg<sub>2</sub>Y, Mg<sub>2</sub>La and Mg<sub>3</sub>La crystal structures is calculated and analyzed, as shown in Fig.3. The contour lines are drawn within the following ranges: Mg<sub>2</sub>Y: -4.232×10<sup>-2</sup> to 7.631×10<sup>-2</sup> v/nm<sup>3</sup>, Mg<sub>2</sub>La: -4.423×10<sup>-2</sup> to 1.367×10<sup>-1</sup> v/nm<sup>3</sup>, Mg<sub>3</sub>La: -4.647×10<sup>-2</sup> to 1.875×10<sup>-1</sup> v/nm<sup>3</sup>. The spacings are: Mg<sub>2</sub>Y, 0.423×10<sup>-2</sup> v/nm<sup>3</sup>, Mg<sub>2</sub>La, 4.524×10<sup>-2</sup> v/nm<sup>3</sup> and Mg<sub>3</sub>La, 5.423×10<sup>-2</sup> v/nm<sup>3</sup>. From the electron density difference, it can be inferred that Mg-Y atoms in Mg<sub>2</sub>Y are ionic bonds , Mg-Mg atoms are covalent bonds and Y-Y atoms are metallic bonds. Mg-La atoms in Mg<sub>2</sub>La and Mg<sub>3</sub>La are ionic bonds, Mg-Mg atoms are covalent bonds, and the La-La atoms are metallic bonds. Based on the

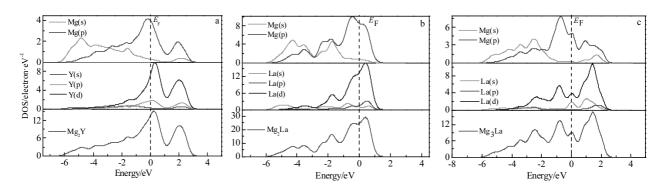


Fig.2 Total (Partial) density of states of Mg<sub>2</sub>Y(a), Mg<sub>2</sub>La (b), and Mg<sub>3</sub>La (c)

Table 4 Electron occupation numbers of $Mg_2Y$ , $Mg_2La$ , and $Mg_3La$						
Phase	Atom	s	р	d	Total	Charge, e
	Mg(I)	0.94	7.02	0.00	7.97	0.03
$Mg_2Y$	Mg(II)	0.96	7.08	0.00	8.04	-0.04
	Y	0.59	0.53	1.82	2.95	0.05
Mala	Mg	0.87	7.23	0.00	8.09	-0.09
Mg <sub>2</sub> La	La	2.60	6.05	2.16	10.81	0.19
	Mg(I)	0.75	7.15	0.00	7.90	0.10
Mg <sub>3</sub> La	Mg(II)	0.87	7.12	0.00	7.98	0.02
	La	2.29	6.32	2.52	11.14	-0.14

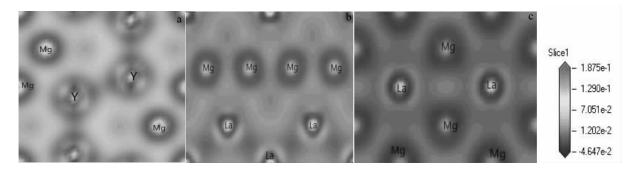


Fig.3 Electron density difference of Mg<sub>2</sub>Y (a), Mg<sub>2</sub>La (b), and Mg<sub>3</sub>La (c)

above analysis, it can be found that the stability of  $Mg_2Y$ ,  $Mg_2La$  and  $Mg_3La$  crystal structures is based on the interaction of ionic bond, covalent bond and metallic bond.

# 2.3 Elastic properties

The ability of a material to resist external forces is generally analyzed by elastic constants. Mg<sub>2</sub>Y has a hexagonal crystal structure with five independent elastic constants  $C_{11}$ ,  $C_{12}$ ,  $C_{13}$ ,  $C_{33}$  and  $C_{44}$ , satisfying the following stability criteria<sup>[25, 26]</sup>:  $C_{11}$ >0,  $C_{11} - C_{12}$ >0,  $C_{44}$ >0,  $(C_{11}+C_{12})C_{33} - 2C_{33}^2$ >0. Mg<sub>2</sub>La and Mg<sub>3</sub>La are cubic crystal structures with three independent elastic constants  $C_{11}$ ,  $C_{12}$  and  $C_{44}$ , satisfying the following stability criteria<sup>[27]</sup>:  $(C_{11}+2C_{12})/3>0$ ,  $C_{11} - C_{12}>0$ ,  $C_{44}>0$ . The calculated elastic constants of Mg<sub>2</sub>Y, Mg<sub>2</sub>La and Mg<sub>3</sub>La are shown in Table 5. All the three crystal structures meet the stability criteria, indicating that they all have a stable structure, which is consistent with the conclusion obtained by the cohesive energy analysis. The elastic modulus of the polycrystals can be obtained from the results of the elastic constants of the single crystals. For Mg<sub>2</sub>La and Mg<sub>3</sub>La with cubic crystal structures, the bulk modulus *B* and shear modulus *G* can be obtained by the following equation <sup>[28]</sup>.

$$B = \frac{1}{3} \left( C_{11} + 2C_{12} \right) \tag{3}$$

$$G = \frac{1}{5} \left( 3C_{44} + C_{11} - C_{12} \right) \tag{4}$$

For hexagonal Mg<sub>2</sub>Y, the bulk modulus *B* and the shear modulus *G* are calculated based on the VRH approximation<sup>[10, 29]</sup>. Voigt approximation equation is as follows:

$$B_{\rm v} = \frac{1}{9} \Big[ 2 \left( C_{11} + C_{12} \right) + C_{33} + 4 C_{13} \Big]$$
(5)

$$G_{\rm v} = \frac{1}{30} \left( 7C_{11} - 5C_{12} + 12C_{44} + 2C_{33} - 4C_{13} \right) \tag{6}$$

n	hasa		Elastic constants/GPa						
Phase -		$C_{11}$	$C_{12}$	$C_{13}$	$C_{33}$	$C_{44}$			
	Present	79.18	25.44	27.84	81.84	19.11			
$Mg_2Y$	Cal. [11]	76.83	25.41	21.11	83.96	17.78			
	Cal. [8]	62.86	39.91	27.16	77.07	17.42			
	Present	52.24	25.65	-	-	20.80			
Mg <sub>2</sub> La	Cal. [21]	58.00	25.00	-	-	22.20			
	Cal. [27]	58.17	25.03	-	-	20.32			
	Present	60.96	29.65	-	-	34.24			
Mg <sub>3</sub> La	Cal. [18]	57.91	27.10	-	-	36.18			
-	Cal. [9]	56.47	26.83	-	-	22.08			

Table 5 Elastic constants of Mg<sub>2</sub>Y, Mg<sub>2</sub>La and Mg<sub>3</sub>La

Reuss approximation formula is as follows:

$$B_{\rm R} = \frac{c^2}{C_{11} + C_{12} + 2C_{33} - 4C_{13}} \tag{7}$$

$$G_{\rm R} = \frac{5}{2} \left\{ \frac{c^2 C_{44} C_{66}}{3 B_{\rm V} C_{44} C_{66} + c^2 (C_{44} + C_{66})} \right\}$$
(8)

Hill approximation formula is as follows:

$$B = \frac{1}{2}(B_{\rm V} + B_{\rm R})$$
(9)

$$G = \frac{1}{2}(G_{\rm V} + G_{\rm R}) \tag{10}$$

The Young's modulus E, the Poisson's ratio v and the elastic anisotropy coefficient A can be obtained from the results of the bulk modulus and the shear modulus, by using the following equation <sup>[8, 17]</sup>:

$$E = \frac{9BG}{3B+G} \tag{11}$$

$$\nu = \frac{1}{2} \left[ \frac{B - (2/3)G}{B + (1/3)G} \right]$$
(12)

$$A = \frac{2C_{44}}{C_{11} - C_{12}} \tag{13}$$

The calculated results of the elastic modulus and the Poisson's ratio of  $Mg_2Y$ ,  $Mg_2La$  and  $Mg_3La$  are shown in Table 6. It can be seen that the calculated results are consistent with the results listed in other literatures, indicating that the calculation method is reliable. It is found that the bulk modulus *B* of the three crystal structures is larger than the shear modulus *G*, indicating that the stability of the structure is mainly affected by the shear modulus *G*.

Modulus *B* is usually used to characterize the ability of a material to resist deformation under applied stress<sup>[31]</sup>. The greater the value of the bulk modulus *B*, the stronger the resistance to deformation. It can be seen from Table 6 that the resistance ability of the three kinds of compounds to extend force deform is:  $Mg_2Y > Mg_3La > Mg_2La$ . Similarly, the shear modulus *G* can be used to characterize the ability of a material to resist shear deformation under shear stress<sup>[31]</sup>. The larger the value of shear modulus, the stronger the resistance to shear deformation. It can be seen from Table 6 that the resistance ability of the three kinds of shear modulus of shear modulus.

compounds to shear deformation is Mg<sub>3</sub>La>Mg<sub>2</sub>Y>Mg<sub>2</sub>La. Young's modulus *E* can be used as a measure of solid stiffness. The greater the Young's modulus *E*, the greater the stiffness of the material<sup>[31, 32]</sup>. The stiffness of the three compounds is Mg<sub>3</sub>La>Mg<sub>2</sub>Y>Mg<sub>2</sub>La. Poisson's ratio is a measure of the resistance of a crystal to shear, usually the value is between -1 and 0.5. The greater the Poisson's ratio, the better the plasticity of a material<sup>[31]</sup>. The Poisson's ratio result in Table 6 indicates that plasticity order from strong to weak is Mg<sub>2</sub>La>Mg<sub>2</sub>Y>Mg<sub>3</sub>La. The Poisson's ratio between 0.25 and 0.5 means that the binding force among the atoms in the material is the central atomic force. The Poisson's ratio of Mg<sub>2</sub>Y and Mg<sub>2</sub>La is between 0.25 and 0.5, indicating that their interatomic binding energies are all central atomic forces.

The ratio of shear modulus to bulk modulus can be used to evaluate the material ductility. If the *G/B* ratio is less than 0.57, then the material shows ductility, otherwise the material is brittle. The *G/B* values of Mg<sub>2</sub>Y, Mg<sub>2</sub>La and Mg<sub>3</sub>La are 0.522, 0.504 and 0.624, respectively, indicating that Mg<sub>2</sub>Y and Mg<sub>2</sub>La are ductile phases and Mg<sub>3</sub>La is a brittle phase. In addition, the ductility and brittleness of a material also can be estimate by the values of the elastic constants  $C_{12}-C_{44}^{[33]}$ . If  $C_{12}-C_{44}<0$ , the material is brittle, whereas the material is ductile. As can be seen from Table 6 that the  $C_{12}-C_{44}$  values of Mg<sub>2</sub>Y and Mg<sub>2</sub>La are positive, implying that they are ductile material. Whereas the  $C_{12}-C_{44}$  value of Mg<sub>3</sub>La is negative implying that it is a brittle material, which is in agreement with the result estimated by *G/B* value.

#### 2.4 Melting temperature and hardness

It is well known that the properties of materials are closely associated with the hardness and melting temperature. The hardness and melting temperature are important indicators to evaluate the abrasive resistance and heat resistance of materials<sup>[34]</sup>. Here, the hardness and melting temperature of Mg<sub>2</sub>Y, Mg<sub>2</sub>La and Mg<sub>3</sub>La are further studied. The hardness *H* can be calculated based on the following semi-empirical equation <sup>[35]</sup>:

$$H = \frac{(1-2\nu)E}{6(1+\nu)}$$
(14)

P	hase	B/GPa	G/GPa	E/GPa	v	A	G/B	( <i>C</i> <sub>12</sub> - <i>C</i> <sub>44</sub> )/GPa
	Present	44.70	23.32	59.60	0.2778	0.710	0.522	6.33
Mg <sub>2</sub> Y	Cal. [17]	41.43	23.59	59.48	0.2607	0.69	0.569	7.63
	Cal. [30]	42.00	23.00	59.00	0.2640	-	0.548	-
Mg <sub>2</sub> La	Present	34.51	17.38	44.65	0.2844	1.56	0.504	4.85
	Cal. [21]	36.00	19.80	49.90	-	-	0.550	2.80
	Cal. [27]	36.08	16.57	47.90	-	1.23	0.459	4.71
Mg <sub>3</sub> La	Present	40.09	25.01	62.11	0.2418	2.19	0.624	-4.59
	Cal. [18]	37.37	25.69	62.70	0.2200	2.34	0.687	-9.08
	Cal. [21]	37.40	26.50	64.30	-	-	0.709	-10.1

 Table 6
 Calculated bulk modulus (B), shear modulus (G), Young's modulus (E), Poisson ratio (v), anisotropic coefficient (A) and

 G/B of Mg<sub>2</sub>Y, Mg<sub>2</sub>La and Mg<sub>3</sub>La

For metal materials, the melting temperature  $T_{\rm m}$  can also be estimated using the  $C_{11}$  elastic constant in intermetallic compounds, as given below <sup>[36]</sup>:

 $T_{\rm m} = 553 \,\mathrm{K} + (5.91 \,\mathrm{K}/\mathrm{GPa}) C_{11} \pm 300 \,\mathrm{K}$  (15)

The calculated results of hardness and melting temperature are shown in Table 7. It can be seen that  $Mg_3La$  has the highest hardness in the three phases followed by  $Mg_2Y$  and  $Mg_2La$ . It can also be seen that  $Mg_2Y$  has the highest melting temperature in the three phases followed by  $Mg_3La$  and  $Mg_2La$ . These calculated results indicate that  $Mg_3La$  has the best abrasive resistance while  $Mg_2Y$  has the best heat resistance among the three compounds.

 Table 7
 Hardness and melting temperature parameters of Mg2Y, Mg2La and Mg3La

8 / 8	0	
Phase	H/GPa	$T_{\rm m}/{ m K}$
$Mg_2Y$	5.64	1021
Mg <sub>2</sub> La	4.12	862
Mg <sub>3</sub> La	6.64	913

# 3 Conclusions

1) The calculated crystal structures of  $Mg_2Y$ ,  $Mg_2La$  and  $Mg_3La$  are in good agreement with literature values. The analysis of formation and cohesive energy show that the  $Mg_3La$  has stronger alloying ability and  $Mg_2La$  has stronger structural stability. The addition of Y and La can improve the stability of Mg-Zn alloy.

2) The analysis of the electronic structure show that  $Mg_2La$  has the strongest conductivity and stability, followed by  $Mg_2Y$  and  $Mg_3La$ . The stability of  $Mg_2Y$ ,  $Mg_2La$  and  $Mg_3La$  crystal structures are determined by the interaction of ionic bond, covalent bond and metal bond.

3) The elastic constants show that three phases are mechanically stable. The analyses of bulk modulus B, shear modulus G, Young's modulus E and Poisson's ratio v show that Mg<sub>2</sub>Y has the strongest resistance to deformation, Mg<sub>3</sub>La has the strongest resistance to shear deformation and stiffness,  $Mg_2Y$  and  $Mg_2La$  are ductile phase,  $Mg_3La$  is a brittle phase.

4) The calculated results of hardness and melting temperature show that  $Mg_3La$  has the largest hardness and  $Mg_2Y$  has the highest melting temperature.

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# Mg-Zn-Y-La 合金中二元相的电子结构和力学性质的第一性原理计算

高 岩<sup>1,2</sup>,毛萍莉<sup>1</sup>,刘 正<sup>1</sup>,王 峰<sup>1</sup>,王 志<sup>1</sup>
(1. 沈阳工业大学,辽宁 沈阳 110870)
(2. 沈阳师范大学,辽宁 沈阳 110034)

摘 要:为了研究稀土元素对镁锌合金性能的影响,利用基于第一性原理计算的平面波赝势方法,对 Mg<sub>2</sub>Y、Mg<sub>2</sub>La 和 Mg<sub>3</sub>La 的 结构稳定性、电子结构和力学性能进行了计算和分析。形成热和结合能的计算结果表明,Mg<sub>3</sub>La 具有最强合金化能力,而 Mg<sub>2</sub>La 具有最强的结构稳定性。通过电子态密度(DOS),电子占据数和差分电荷密度分析了结构的稳定机制。计算了 3 种结构的弹性常数,并进一步得到了体模量 *B*,剪切模量 *G*,杨氏模量 *E* 和泊松比 y 等。计算结果表明:Mg<sub>2</sub>Y 具有最强的抵抗变形能力,Mg<sub>3</sub>La 具有最强的刚度和抵抗剪切变形能力,而 Mg<sub>2</sub>La 塑性最强。进一步分析表明 Mg<sub>2</sub>Y 和 Mg<sub>2</sub>La 为延性相,而 Mg<sub>3</sub>La 为脆性相。此外,硬度和熔点的计算结果表明,3 种金属间化合物中,Mg<sub>3</sub>La 的硬度最大,Mg<sub>2</sub>Y 的熔点最高。 关键词:镁合金;第一性原理计算;结构稳定性;电子结构;弹性性质

作者简介: 高 岩, 男, 1983 年生, 博士, 沈阳工业大学材料科学与工程学院, 辽宁 沈阳 110870, 电话: 024-86593280, E-mail: gaoy@synu.edu.cn

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