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Estimation of Thermodynamic Properties of $Li_{1+x}M_{1-x}O_2$ Cathode Material for Lithium-Ion Battery (Lithium-Rich) Based on the Group Contribution Method

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Abstract: Given the lack of thermodynamic data on $\text{Li}_{1+x}M_{1-x}O_2$ materials, LiAlO_2 was split in accordance with the principle of the group contribution method. Mathematical models for estimating the $\Delta G^{\theta}_{f,298}$, $\Delta H^{\theta}_{f,298}$, and C_p of LiAlO_2 were proposed on the basis of thermodynamic principles. The group contribution method was used to estimate the $\Delta G^{\theta}_{f,298}$, and $\Delta H^{\theta}_{f,298}$ of 56 solid inorganic compounds and the $C_{p,298}$ of 54 solid inorganic compounds to test the reliability and applicability of the model. The group contribution method was used to estimate the associate the experimental data selected by fitting group parameters are accurate and reliable, and the group division method is appropriate. Mathematical models for estimating the $\Delta G^{\theta}_{f,298}$, $\Delta H^{\theta}_{f,298}$, $\Delta H^{\theta}_{f,298}$, and C_p of three types of $\text{Li}_{1+x}M_{1-x}O_2$ materials were constructed on the basis of the satisfactory results. The $\Delta G^{\theta}_{f,298}$, $\Delta H^{\theta}_{f,298}$, and $C_{p,298}$ of the 63 common $\text{Li}_{1+x}M_{1-x}O_2$ materials were also estimated.

Key words: lithium-rich cathode materials; the group contribution method; $\Delta G^{\theta}_{f,298}$; $\Delta H^{\theta}_{f,298}$; $C_{p,298}$

The gradual depletion of traditional fossil energy has led to the emergence of various energy materials and energy conversion devices. Electromagnetic (EM) functional materials support the rapid expansion of modern automation engineering, electronic science and information technology. As representatives of low-dimensional electromagnetic functional materials, silicon carbide, zinc oxide, carbon materials, transition metals and their oxides, metal-organic framework materials, etc, due to their excellent electromagnetic functions, have been used in the fields of communication and imaging, electromagnetic wave absorption and shielding, etc, showing a broad application prospect^[1]. In addition, a lot of work is continuously devoted to the development and update of the collection and conversion of environmental capabilities, such as water wave energy, light energy, and heat energy^[2]. High-performance rechargeable lithium-ion batteries, as a representative of advanced energy storage technology, have made great efforts to develop cathode and anode materials. For example, Cao et

al prepared layered NiCo₂O₄ nanofibers assembled with hollow nanoparticles through simple electrospinning technology and annealing process to improve the electrochemical performance of lithium-ion batteries^[3]. Cao et al^[4] designed a high-capacity electrode with a threedimensional layered Co₃O₄ flower-like structure with mesogenic nanostructures.

Lithium-ion batteries have the characteristics of light weight, small size, low self-discharge rate, long cycle life, and wide operating temperature range while lack of memory effect. Since their inception, lithium-ion batteries have been developed rapidly and widely used in 3C digital consumer electronics products, pure electric vehicles (EVs), hybrid vehicles (HEVs), and plug-in HEVs^[5-7]. The challenges of lithium-ion batteries continue to grow with the increasing functional demands of consumer electronic products and the mounting of EVs. The lithium-ion battery cathode material LiCoO₂ and ternary material LiNi_xCo_yMn_zO₂ has a theoretical specific capacity of 272 mAh/g, but it is prone to irreversible

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phase change due to excessive delithiation in actual use. Generally, the maximum deintercalation of lithium accounts for 60%~80% of the total content. Thus, the actual specific capacity of the lithium-ion battery cathode material is less than 200 mAh/g, which is lower than the anode capacity of commercial carbon electrodes (350 mAh/g). In turn, the anode capacity of commercial carbon electrodes is considerably lower than that of silicon and tin anode electrodes. Given that the energy density of lithium-ion batteries depends mainly on the specific capacity of their cathode materials, researching and developing new lithium-ion battery cathode materials with increased specific capacity is necessary to meet the development requirements of EVs and HEVs in the future^[8,9].

Lithium-ion battery cathode materials currently on the market are mainly classified as (1) hexagonal layered LiCoO_2 , LiNiO_2 , and $\text{LiNi}_x\text{Co}_y\text{Mn}_2\text{O}_2$ structures with R-3m spatial point groups; (2) spinel-structured LiMn_2O_4 with Fd-3m spatial point groups; and (3) compound LiFePO_4 with a polyanionic structure. The difference in the spatial structure determines the lithium ion adsorption behavior and electrochemical performance of the two-dimensional material, and from the micro level, the basic relationship between the electronic structure and its electrochemical properties is that the electrical conductivity of the material and the adsorption strength of the intermediate are the key factors determining the electrocatalytic kinetics^[10].

Lithium-rich cathode materials ($Li_{1+x}M_{1-x}O_2$, 0<x<1, M=Ni, Co, Mn...) have an α -NaFeO₂ structure and belong to the hexagonal crystal system and the R-3m space group. Li occupies the 3a position, the transition metal M occupies the 3b position, and O occupies the 6c position. The transition metal layer Li, transition metal M, and the structure of the transition metal layer are similar to those of $LiCoO_2^{[11-13]}$. The lithium-rich $Li_{1+r}M_{1-r}O_2$ cathode material is a solid solution that is composed of different proportions of LiMO2 and Li₂MnO₂ materials, namely xLi_2 MnO₂ $(1-x)LiMO_2$ $(0 \le x \le 1, M$ =Ni, Co, Cr, Fe, Al, Mg, Ni-Co, Ni-Mn, Ni-Co-Mn...), if the cation ordering problem is ignored. LiMO, has an α -NaFeO, layered structure and belongs to the R-3m space group. Transition metal ions occupy octahedral positions to form the MO_6 layer in Li MO_2 ; by contrast, and the structure in the Li layer is tetrahedral^[14,15]. Li₂MnO₃ belongs to the monoclinal crystal system and the C2/m spatial group and is a layered rock salt structure. Li⁺ and Mn⁴⁺ in Li₂MnO₃ are arranged alternately at the molar ratio of 1:2 to form a transition metal layer $[Li_{1/3}Mn_{2/3}]$. The superlattice structure is ascribed to the mixing effect of Li⁺ and Mn⁴⁺ present between transition metal layers and is reflected as a sawtooth peak in the 2θ range of 20°~28° in XRD patterns. The lithium-rich cathode material is also considered as a transition metal moiety that replaces Li⁺ and Mn4+ in the Li[Li Mn]O transition metal layer, which is expressed as $\text{Li}[M_1, \text{Li}_{2}Mn_{2\times 2}]O_2^{[16,17]}$. The valence states of the transition metals are Ni^{2+} , Co^{3+} and Mn^{4+} , and the three elements have different effects on material electrochemical properties. Mn4+ can improve material structural stability, thermal stability, and safety while reducing the costs.

Nevertheless, it lacks electrochemical activity during charging. Excessive Mn^{4+} content will reduce the specific capacity of materials and subsequently facilitate spinel phase production and destroy the layered material structure. Although Ni²⁺ is helpful for improving charging and discharging capacity, excessive Ni²⁺ will cause a miscible effect in combination with Li⁺. This miscible effect will result in poor cycling performance and multiplier performance. Co³⁺ can effectively inhibit cationic mixing and improve electronic conductivity and cycle performance of the material. However, increasing Co³⁺ ratio will reduce capacity^[18-21]. Fig. 1 shows the schematic of the layered Li₂MnO₃ and LiMO₂ structures, where *M* is a transition metal, such as Ni, Co, or Mn.

The initial discharge specific capacity of the lithium-rich material in the high voltage state can exceed 300 mAh/g, which is much higher than that of currently commercially available cathode materials. However, layered lithium-rich materials cannot meet application requirements because they exhibit several problems, such as large initial irreversible capacity loss, voltage attenuation during cycling, low tap density, and poor rate performance^[22,23]. These deficiencies can be improved through element doping and surface coating. Layered lithium-rich materials are now considered as ideal cathode materials for next-generation lithium-ion batteries with the development of the synthesis and modification technology of lithium-rich materials, because of their high specific capacity, excellent cycling ability and novel electrochemical charging and discharging mechanism^[24-26].

Current research on lithium-rich $Li_{1+x}M_{1-x}O_2$ materials is mainly focused on their structure and electrochemical properties and little attention is paid on their thermodynamic properties. The development of novel lithium-rich $Li_{1+r}M_{1-r}O_2$ materials with high energy density and capacity is dependent on the internal relationship among their structures, thermodynamic properties, and electrochemical properties. It is specifically manifested with changing the crystal phase of the material by adjusting the different charge distribution and electronic structure of various atomic configurations. The phase change process in the chemical reaction involved in this process requires reliable thermodynamic data to guide. It also provides a theoretical reference for estimating actual battery energy density and theoretical capacity. Other issues include the further improvement of the charging and discharging performances of lithium-ion batteries, the development of new



Fig.1 Structural schematic diagram of layered Li_2MnO_3 (a) and $LiMO_2$ (b) (*M* is Ni, Co, or Mn)

electrode materials and battery systems, and the optimization of battery structure. The limited understanding of the thermodynamic properties of $\text{Li}_{1+x}M_{1-x}\text{O}_2$ restricts the development and utilization of new lithium-rich materials. The study of the thermodynamic properties of $\text{Li}_{1+x}M_{1-x}\text{O}_2$ are crucial for guiding theoretical calculations toward the improvement and usage of lithium-rich new materials in the long run.

Mostafa et al^[27,28] proposed that the thermodynamic properties of solid inorganic compounds are the sum of the thermodynamic group contributions of their constituent cations, anions, and ligands (such as water of crystallization). The $\Delta H^{\theta}_{f,298}$ values of 938 solid inorganic compounds and the $\Delta G^{\theta}_{f,298}$ values of 687 solid inorganic compounds were subjected to multiple linear regression analysis (84% of the data are published by the International Union of Pure and Applied Chemistry). The group contribution values of 136 cations, 23 anions, and 5 ligand molecules were obtained, and the relative errors of each group contribution value were estimated by the standard deviation of the parameters. The $C_{p,298}$ values of 664 solid inorganic compounds were also subjected to multiple linear regression analysis. The group contribution values of 129 cations, 17 anions, and 2 ligands were acquired. The group contribution method has a wide range of applications and high reliability and accuracy. Thus, the mathematical models for estimating the $\Delta H^{\theta}_{f,298}$, $\Delta G^{\theta}_{f,298}$, and $C_{\rm p}$ of three kinds of lithium-rich ${\rm Li}_{1+r}M_{1-r}O_2$ materials were constructed for the first time by the group contribution method. Moreover, the $\Delta H^{\theta}_{\rm f,298}$, $\Delta G^{\theta}_{\rm f,298}$ and $C_{\rm p,298}$ of 63 common lithium-rich $Li_{1+x}M_{1-x}O_2$ materials were estimated.

1 Estimation Method

1.1 Mathematical model for estimating the thermodynamic data of matter using the group contribution method

The success of the group contribution method in estimating material property data is associated with the physical basis of the mathematical model related to group parameters and properties, the accuracy and reliability of the experimental data selected for fitting group parameters, and the appropriateness of the group partitioning method. LiAlO₂ was split in accordance with the principle of the said method to determine the reliability and application range of the group contribution method for estimating the $\Delta H^{\theta}_{f,298}$, $\Delta G^{\theta}_{f,298}$ and $C_{p,298}$ of solid inorganic compounds. Moreover, the mathematical models for estimating the $\Delta H^{\theta}_{f,298}$, $\Delta G^{\theta}_{f,298}$ and C_{p} of LiAlO₂ were proposed in accordance with thermodynamic principles.

Table 1 shows the group contribution values of Li⁺, Ni²⁺, Ni³⁺, Co²⁺, Co³⁺, Mn³⁺, Mn⁴⁺, and O²⁻ to the $\Delta H^{\theta}_{f,298}$ and $\Delta G^{\theta}_{f,298}$ of the solid inorganic compounds. The $\Delta H^{\theta}_{f,298}$ and $\Delta G^{\theta}_{f,298}$ of LiAlO₂ were estimated by the group contribution method, as follows. (1) The molecular formula of the solid inorganic compound was written. (2) The molecular formula was deconstructed into the structural groups of cations, anions, or

ligands. The contribution value of each molecular structure group was multiplied by the distribution number of the group in the molecular formula. Then, the contribution value of each group was calculated. (3) The mathematical model for estimating the thermodynamic properties of the substance was established. Table 2 presents the estimation process and results.

The mathematical models for estimating the $\Delta H^{\theta}_{f,298}$ and $\Delta G^{\theta}_{f,298}$ of LiAlO₂ by the group contribution method are as follows:

$$\Delta H^{\theta}_{f,298}(\text{LiAlO}_2) = \Delta H^{\theta}_{f,298}(\text{Li}^+) + \Delta H^{\theta}_{f,298}(\text{Al}^{3+}) + 2\Delta H^{\theta}_{f,298}(\text{O}^{2-})$$
(1)

 $\Delta G^{\theta}_{f,298}(\text{LiAlO}_2) = \Delta G^{\theta}_{f,298}(\text{Li}^+) + \Delta G^{\theta}_{f,298}(\text{Al}^{3+}) + 2\Delta G^{\theta}_{f,298}(\text{O}^{2-}) \quad (2)$ in which 1, 1, and 2 are the distribution numbers of Li⁺, Al³⁺, and O²⁻ in the LiAlO, molecular formula, respectively.

Table 2 shows that the estimated values of the $\Delta H^{\theta}_{f,298}$, and $\Delta G^{\theta}_{f,298}$ of LiAlO₂ are – 1193.37 and – 1129.95 kJ·mol⁻¹, respectively. Compared with the values given in the literature (–1188.70 and –1126.30 kJ·mol⁻¹), it reveals that the corresponding relative errors of the estimated values are 0.39% and 0.32%^[29].

Table 3 shows the group contribution values of Li⁺, Ni²⁺, Ni³⁺, Co²⁺, Co³⁺, Mn³⁺, Mn⁴⁺, and O²⁻ to the C_p of the solid inorganic compound. The group contribution method was used to estimate the C_p of LiAlO₂, as follows. (1) The molecular formula of the solid inorganic compound was written. (2) The formula was broken into the cationic and anionic structural groups given in Table 1, and the contribution number of the group in the molecular formula to calculate the contribution value of each group. (3) The contribution values of $\sum_j n_j \Delta_{a,j}$, $\sum_j n_j \Delta_{b,j}$, $\sum_j n_j \Delta_{c,j}$ and $\sum_j n_j \Delta_{d,j}$ were determined by tallying the contribution values of each group. (4) The mathematical model for estimating the thermodynamic properties of the material was developed. Table 4 presents the estimation process and results.

$$C_{p}(\text{LiAlO}_{2}) = \sum_{j} n_{j} \Delta_{a,j} + (\sum_{j} n_{j} \Delta_{b,j} \times 10^{-3}) T + (\sum_{j} n_{j} \Delta_{c,j} \times 10^{6}) / T^{2} + (\sum_{j} n_{j} \Delta_{d,j} \times 10^{-6}) T^{2}$$

$$\sum_{j} n_{j} \Delta_{a,j} = \Delta_{a,j}(\text{Li}^{+}) + \Delta_{a,j}(\text{Al}^{3+}) + 2\Delta_{a,j}(\text{O}^{2-})$$

$$\sum_{j} n_{j} \Delta_{b,j} = \Delta_{b,j}(\text{Li}^{+}) + \Delta_{b,j}(\text{Al}^{3+}) + 2\Delta_{b,j}(\text{O}^{2-})$$

$$\sum_{j} n_{j} \Delta_{c,j} = \Delta_{c,j}(\text{Li}^{+}) + \Delta_{c,j}(\text{Al}^{3+}) + 2\Delta_{d,j}(\text{O}^{2-})$$

$$\sum_{j} n_{j} \Delta_{d,j} = \Delta_{d,j}(\text{Li}^{+}) + \Delta_{d,j}(\text{Al}^{3+}) + 2\Delta_{d,j}(\text{O}^{2-})$$
(3)

in which 1, 1, and 2 are the distribution numbers of Li^+ , Al^{3+} , and O^{2-} in the LiAlO₂ molecular formula, respectively.

The $C_{p,298}$ of LiAlO₂ was calculated in accordance with the required temperature (with T=298 K as the reference data). The estimated value of $C_{p,298}$ is 67.75 J·mol⁻¹·K⁻¹. Compared with the value of 67.78 J·mol⁻¹·K⁻¹ given in the literature, the relative error of the estimated value is 0.04%.

Kopp's rule states that the constant pressure heat capacity C_p^{θ} of a compound is approximately equal to the sum of the molar heat capacity of the atoms that constitute the elements of the compound, as shown in Eq. (4). Table 5 lists the contribution values of elements, such as Ca, Al, Li, Ni, Co, Mn, and O^[30].

$$C_{p}^{\theta} = \sum_{j} n_{j} C_{p,j}^{\theta}$$
(4)

Table 1	Solid inorganic compound groups of cationic, anionic and ligand molecule groups, their contributions $(\Delta_{H,j}, \Delta_{G,j})$, number o
	occurrences in the regression, and standard deviations $(S_{_{\Delta j}})$ associated with them for standard heats and free energies o
	formation ^[27]

	No. of occurrence		A /kI.mol ⁻¹ S f	S for A	No. of occurrence	4 //r I.mol ⁻¹	S for A
	Gloup	for $\Delta_{H,j}$	$\Delta_{H,j}/KJ$ mor	$S_{\Delta j}$ for $\Delta_{H,j}$	for $\Delta_{G,j}$	Δ_{GJ} KJ IIIOI	$\mathcal{D}_{\Delta j}$ for $\Delta_{G,j}$
	Al^{3+}	129	-553.115	12.713	118	-420.023	23.951
	B^{3+}	50	-392.895	13.015	49	-273.681	24.327
	Be^{2^+}	23	-402.267	13.713	15	-297.827	19.827
	Ca^{2+}	131	-520.898	14.734	124	-432.414	16.222
	Co^{2^+}	23	-76.275	30.537	3	4.128	35.818
	Co^{3^+}	1	-37.950	-	1	94.826	-
	Cr^{3^+}	15	-290.075	27.492	15	-168.532	25.868
	Fe^{2+}	40	-126.475	13.177	22	-43.568	19.883
	Fe^{3+}	30	-122.815	14.811	22	-8.638	25.237
	Hf^{4+}	1	-769.852	-	1	-606.500	-
	Li^{+}	42	-292.950	8.187	29	-250.254	11.449
	Mg^{2+}	116	-461.804	9.478	72	-372.414	16.624
Cation	Mn^{2^+}	16	-250.029	14.052	13	-167.203	19.858
	$\mathrm{Mn}^{\mathrm{3+}}$	1	-167.921	-	1	-100.187	-
	$\mathrm{Mn}^{^{4+}}$	1	-85.813	-	1	66.580	-
	Mo^{6^+}	38	-234.996	25.436	82	-17.811	47.790
	NH_4^+	8	-164.486	16.381	5	-53.199	20.031
	Ni ²⁺	20	-91.381	14.364	14	0.357	20.625
	Ni ³⁺	2	-20.042	31.141	1	149.483	-
	P^{5+}	4	-481.148	29.622	4	-285.069	44.693
	Si ⁴⁺	174	-575.556	16.706	159	-411.036	32.072
	Ti ⁴⁺	15	-571.358	22.030	17	-408.836	34.023
	V^{5+}	32	-369.937	22.053	33	-162.620	40.397
	W^{6+}	40	-367.612	26.321	40	-131.422	48.219
	Zn^{2+}	33	-191.646	11.753	32	-91.944	17.855
	CO32-	26	-616.496	14.214	28	-635.990	19.167
	F ⁻	291	-310.918	4.638	168	-326.667	8.264
Anion	NO ₃ ⁻	97	-175.710	6.206	40	-126.405	10.288
AIIIOII	O ²⁻	1946	-173.650	8.264	1725	-229.836	15.876
	PO ₄ ³⁻	22	-1197.521	16.826	20	-1206.614	26.773
	SO4 ²⁻	121	-814.755	9.861	91	-795.046	16.975
Ligand molecule	H ₂ O	718	-298.933	1.133	394	-244.317	1.470

Table 2 Estimation of the $\Delta H^{\theta}_{f,298}$, and $\Delta G^{\theta}_{f,298}$ of LiAlO₂ (kJ·mol⁻¹)

(,	
Group	$\Delta H^{ heta}_{ m f,298}$	$\Delta G^{ heta}_{ m f,298}$
Li^{+}	-292.50	-250.254
Al ³⁺	-553.115	-420.023
20 ²⁻	2×(-173.650)	2×(-229.836)
Total	-1193.37	-1129.95

The $C_{p,298}$ of LiAlO₂ was estimated by Kopp's rule through the following steps. (1) The molecular structure formula of the solid inorganic compound was written. (2) The molecular structure formula was broken down in accordance with the types of elements, and the contribution values of the elements were ascertained from Table 5. The contribution value of the element was multiplied by the distribution number of the component in the molecular structure equation, and the contribution value of each element was calculated. (3) The estimated and reported values of $C_{\rm p,298}$ are 68.16 and 67.78 J·mol⁻¹·K⁻¹, respectively. Thus, the relative error of the estimated value is 0.56%. Table 6 presents the estimation process and results.

1.2 Verification of the model

The group contribution method was used to estimate the $\Delta H^{\theta}_{f,298}$ and $\Delta G^{\theta}_{f,298}$ of 56 solid inorganic compounds to determine the reliability and applicability of the model. The

Table 3Solid inorganic salt groups of cationic, anionic groups
and their contribution $(\Delta_{a,j}, \Delta_{b,j}, \Delta_{c,j}, \Delta_{d,j})$ and number of
occurrences in the regression
[28]

	Group	No. of	٨	٨	٨	۸	
	Oloup	occurrence	$\Delta_{\mathbf{a},j}$	$\Delta_{\mathbf{b},j}$	$\Delta_{\mathbf{c},j}$	└ ` d,j	
	Al^{3+}	81	10.306	4.518	-0.623	-3.701	
	B^{3+}	59	-13.188	16.765	0.273	-0.219	
	Be^{2^+}	16	-5.164	22.314	-0.002	6.544	
	Ca^{2+}	64	20.470	-6.225	-0.026	-3.219	
	Co^{2^+}	17	26.552	-19.272	0.327	5.245	
	Co^{3^+}	1	34.158	-42.075	-0.149	-4.614	
	Cr^{3+}	22	21.086	-16.392	0.067	5.298	
	Fe^{2^+}	15	20.486	-5.415	0.055	5.393	
	Fe^{3+}	26	16.618	3.328	0.082	5.002	
	Hf^{4+}	9	20.580	-22.025	-0.050	2.633	
	Li^+	47	15.639	4.124	-0.007	7.973	
	Mg^{2+}	32	14.639	-0.637	-0.074	-0.609	
Cation	Mn^{2+}	15	21.419	-15.908	0.265	5.652	
	Mn^{3+}	4	8.913	3.070	0.454	5.448	
	Mn^{4+}	1	-3.593	22.048	0.643	5.244	
	Mo^{6^+}	11	-7.473	-7.423	1.279	13.430	
	Ni ²⁺	14	22.497	-6.671	-0.022	6.234	
	Ni ³⁺	1	3.549	-7.964	0.413	7.913	
	P^{5+}	12	-39.486	50.248	1.686	2.325	
	Si^{4+}	79	-2.308	4.382	-0.041	-3.301	
	Ti ⁴⁺	34	10.043	-7.562	0.299	11.414	
	V^{5+}	20	-1.279	-5.243	0.697	13.608	
	W^{6+}	20	-23.788	35.451	1.836	9.893	
	Zn^{2+}	18	12.599	-0.744	0.377	4.203	
	Zr^{4+}	13	17.188	-23.478	-0.063	7.098	
	CO ₃ ²⁻	15	47.278	86.757	-0.887	-5.133	
	F^-	247	22.041	15.652	-0.244	1.538	
Anion	NO_3^{-}	120	49.766	83.928	-0.478	-7.040	
AIII0II	O ²⁻	1155	28.152	12.043	-0.747	-4.023	
	PO ₄ ³⁻	6	95.827	56.863	-2.462	-7.691	
	SO_4^{2-}	49	85.866	52.357	-1.925	-0.047	

Table 4	Estimation	of $C_{n,298}$	for LiAlO,	
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Group	$\Delta_{\mathbf{a},j}$	$\Delta_{\mathrm{b},j}$	$\Delta_{\mathrm{c},j}$	$\Delta_{\mathrm{d},j}$			
Li^+	15.639	4.124	-0.007	7.973			
Al^{3+}	10.306	4.518	-0.623	-3.701			
20 ²⁻	2×28.152	2×12.043	2×(-0.747)	2×(-4.023)			
$\sum_{j} n_{j} \Delta_{i,j}$	82.249	32.728	-2.124	-3.774			
$\frac{C_{p,298} = \sum_{n} n \Delta_{a,j}^{+} (\sum_{j} n_{j} \Delta_{b,j}^{+} \times 10^{-3}) T^{+} (\sum_{j} n_{j} \Delta_{c,j}^{-} \times 10^{6}) / T^{2} + (\sum_{j} n_{j} \Delta_{d,j}^{-} \times 10^{-6}) T^{2}}{= 82.249 + (32.728 \times 10^{-3} \times 298 + (-2.124 \times 10^{6})/298^{2} + (-3.774 \times 10^{-6} \times 298^{2}) = 67.75 \text{ J} \cdot \text{mol}^{-1} \cdot \text{K}^{-1}}$							

estimated and reported values of $\Delta H^{\theta}_{f,298}$ were compared^[31]. The absolute values of relative errors were within 4%, and the absolute values of the relative errors of 43 kinds of solid

inorganic compounds were less than 2%. The estimated and reported values of ΔG^{θ}_{f298} were also compared. The absolute values of relative errors were within 4%, the absolute values of the relative errors of 42 kinds of solid inorganic compounds were less than 2%. The $C_{p,298}$ values of 54 solid inorganic compounds were likewise estimated^[32]. The absolute values of the relative errors of $C_{n^{298}}$ were within 5% of the values in the literature. The absolute values of the relative errors of 43 kinds of solid inorganic compounds were less than 2%. The $C_{n 298}$ values of 54 selected solid inorganic compounds were estimated in accordance with Kopp's rule. The absolute values of relative errors of $C_{p,298}$ were within 11% of those reported in the literature, and the absolute values of the relative errors of 33 solid inorganic compounds were less than 3%. Among them, the absolute values of the relative errors of 12 kinds of solid inorganic compounds were 3%~5%. This method has a better calculation effect than Kopp's rule estimation. The results show that the mathematical model of the correlation between group parameters and physical properties is correct, the experimental data selected by fitting the group parameters are accurate and reliable, and the group division method is appropriate.

2 Results and Discussion

Three main types of lithium-rich cathode materials are currently available: (1) lithium-rich layered xLi_2MnO_3 · $(1-x)LiMO_2$ (0<x<1, M=Ni, Co, Cr, Fe, Al, Mg, Ni-Co, Ni-Mn, Ni-Co-Mn...) cathode material; (2) lithium-rich manganese-based $xLi[Li_{1/3}Mn_{2/3}]O_2$ · $(1-x)LiMO_2$ (0<x<1, M=Ni, Co, Cr, Fe, Al, Mg, Ni-Co, Ni-Mn, Ni-Co-Mn...) solid solution material; (3) layered spinel $xLi[Li_{1/3}Mn_{2/3}]O_2$ · $(1-x)LiMn_2O_4$ (0<x<1) composite solid solution material and layered spinel $xLi[Li_{1/3}Mn_{2/3}]O_2$ · $(1-x)LiMn_2O_4$ (0<x<1) composite solid solution material and layered spinel $xLi[Li_{1/3}Mn_{2/3}]O_2$ (1 – x)LiNi_{0.5}Mn_{1.5}O₄ (0<x<1) composite solid solution material. The mathematical models for estimating $\Delta H_{f_{2298}}^{\theta}$, and $\Delta G_{f_{2298}}^{\theta}$, and $\Delta G_{f_{2298}}^{\theta}$ and $C_{p,298}$ of the common 63 lithium-rich materials were estimated.

2.1 Mathematical models for estimating $\Delta H^{\theta}_{f,298}$, $\Delta G^{\theta}_{f,298}$, and C_{p} of the lithium-rich layered $x \text{Li}_{2}\text{MnO}_{3}$. (1-x)LiMO₂ cathode materials

The chemical valences of Li, O, Mn, and *M* in the lithiumrich layered xLi_2MnO_3 \cdot $(1-x)LiMO_2$ (0<x<1, *M*=Ni, Co, Cr, Fe, Al, Mg, Ni-Co, Ni-Mn, Ni-Co-Mn...) materials are +1, -2, +4, and +3, respectively. The most studied systems of the lithium-rich xLi_2MnO_3 \cdot $(1-x)LiMO_2$ material are *M*=Co, Fe, and Ni-Mn. Common lithium-rich layered materials include xLi_2MnO_3 \cdot $(1-x)LiCoO_2$, xLi_2MnO_3 \cdot $(1-x)LiFeO_2$, and xLi_2MnO_3 \cdot $(1-x)Li[Ni_{0.5}Mn_{0.5}]O_2$. The mathematical models for estimating $\Delta H^{\theta}_{f,298}$, $\Delta G^{\theta}_{f,298}$, and C_p of lithium-rich layered xLi_2MnO_3 \cdot $(1-x)LiMO_2$ materials using the group contribution method are shown Eq.(5~7), respectively:

$$\Delta H^{\theta}_{f,298}(xLi_{2}MnO_{3}\cdot(1-x)LiMO_{2})=2x\Delta H^{\theta}_{f,298}(Li^{+}) + x\Delta H^{\theta}_{f,298}(Mn^{4+})+3x\Delta H^{\theta}_{f,298}(O^{2-})+(1-x)\Delta H^{\theta}_{f,298}(Li^{+}) + (1-x)\Delta H^{\theta}_{f,298}(M^{3+})+2(1-x)\Delta H^{\theta}_{f,298}(O^{2-})$$
(5)

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Table 5 Contribution of elements to solid inorganic compound $C_{n^{208}}$

							p;200		
Element	New Kopp's	Element	New Kopp's	Flomont	New Kopp's	Flomont	New Kopp's	Flomont	New Kopp's
Element	Solids								
Н	7.56	F	20.92	Κ	28.78	Ni	25.46	Ba	32.37
Li	23.25	Na	26.19	Ca	28.25	Cu	26.92	W	30.87
Be	12.47	Mg	22.69	Ti	27.24	Br	25.36	Hg	27.87
В	10.10	Al	18.07	V	29.36	Sr	28.41	Pb	31.60
С	10.89	Si	17.00	Mn	28.06	Zr	26.82	Р	22.61
Ν	18.74	S	12.36	Fe	29.08	Mo	29.44	Cr	31.42
0	13.42	Cl	24.69	Со	25.71	Ι	25.29	Misc	26.63

Table 6 Estimation of $C_{p,298}$ for LiAlO₂

Step			LiAlO ₂			
1	Element	$\Delta_{\mathbf{a},j}$	Element	$\Delta_{\mathrm{b},j}$	Element	$\Delta_{\mathrm{c},j}$
2	Li	23.25	Al	18.07	0	2×13.42
3	$\sum_{j} n_{j} \Delta_{i,j}$		$68.16 \text{ J} \cdot \text{mol}^{-1} \cdot \text{K}^{-1}$			

$$\begin{split} &\Delta G^{\theta}_{f;298}(x \text{Li}_{2} \text{MnO}_{3} \cdot (1-x) \text{Li}MO_{2}) = 2x \Delta G^{\theta}_{f;298}(\text{Li}^{+}) \\ &+ x \Delta G^{\theta}_{f;298}(\text{Mn}^{4+}) + 3x \Delta G^{\theta}_{f;298}(\text{O}^{2-}) + (1-x) \Delta G^{\theta}_{f;298}(\text{Li}^{+}) \\ &+ (1-x) \Delta G^{\theta}_{f;298}(M^{3+}) + 2(1-x) \Delta G^{\theta}_{f;298}(\text{O}^{2-}) \qquad (6) \\ &C_{p}(x \text{Li}_{2} \text{MnO}_{3} \cdot (1-x) \text{Li}MO_{2}) = \sum_{j} n_{j} \Delta_{a,j} + (\sum_{j} n_{j} \Delta_{b,j} \times 10^{-3}) T \\ &+ (\sum_{j} n_{j} \Delta_{c,j} \times 10^{6}) / T^{2} + (\sum_{j} n_{j} \Delta_{d,j} \times 10^{-6}) T^{2} \sum_{j} n_{j} \Delta_{a,j} \\ &= 2x \Delta_{a,j}(\text{Li}^{+}) + x \Delta_{a,j}(\text{M}^{4+}) + 3x \Delta_{a,j}(\text{O}^{2-}) \\ &+ (1-x) \Delta_{a,j}(\text{Li}^{+}) + (1-x) \Delta_{a,j}(M^{3+}) + 2(1-x) \Delta_{a,j}(\text{O}^{2-}) \sum_{j} n_{j} \Delta_{b,j} \\ &= 2x \Delta_{b,j}(\text{Li}^{+}) + x \Delta_{b,j}(\text{M}^{4+}) + 3x \Delta_{b,j}(\text{O}^{2-}) \\ &+ (1-x) \Delta_{b,j}(\text{Li}^{+}) + (1-x) \Delta_{b,j}(M^{3+}) + 2(1-x) \Delta_{b,j}(\text{O}^{2-}) \sum_{j} n_{j} \Delta_{c,j} \\ &= 2x \Delta_{c,j}(\text{Li}^{+}) + x \Delta_{c,j}(\text{M}^{4+}) + 3x \Delta_{c,j}(\text{O}^{2-}) \\ &+ (1-x) \Delta_{c,j}(\text{Li}^{+}) + (1-x) \Delta_{c,j}(M^{3+}) + 2(1-x) \Delta_{c,j}(\text{O}^{2-}) \sum_{j} n_{j} \Delta_{d,j} \\ &= 2x \Delta_{d,j}(\text{Li}^{+}) + x \Delta_{d,j}(\text{M}^{4+}) + 3x \Delta_{d,j}(\text{O}^{2-}) \\ &+ (1-x) \Delta_{d,j}(\text{Li}^{+}) + (1-x) \Delta_{d,j}(M^{3+}) + 2(1-x) \Delta_{d,j}(\text{O}^{2-}) \sum_{j} n_{j} \Delta_{d,j} \\ &= 2x \Delta_{d,j}(\text{Li}^{+}) + x \Delta_{d,j}(\text{M}^{4+}) + 3x \Delta_{d,j}(\text{O}^{2-}) \\ &+ (1-x) \Delta_{d,j}(\text{Li}^{+}) + (1-x) \Delta_{d,j}(M^{3+}) + 2(1-x) \Delta_{d,j}(\text{O}^{2-}) \\ &+ (1-x) \Delta_{d,j}(\text{Li}^{+}) + (1-x) \Delta_{d,j}(\text{O}^{2-}) \\ &+ (1-x) \Delta_{d,j}(\text{Li}^{+}) + (1-x) \Delta_{d,j}(M^{3+}) + 2(1-x) \Delta_{d,j}(\text{O}^{2-}) \\ &+ (1-x) \Delta_{d,j}(\text{Li}^{+}) + (1-x) \Delta_{d,j}(M^{3+}) + 2(1-x) \Delta_{d,j}(\text{O}^{2-}) \\ &+ (1-x) \Delta_{d,j}(\text{Li}^{+}) + (1-x) \Delta_{d,j}(\text{O}^{2-}) \\ &+ (1-x) \Delta_{d,j}(\text{Li}^{+}) + (1-x) \Delta_{d,j}(M^{3+}) + 2(1-x) \Delta_{d,j}(\text{O}^{2-}) \\ &+ (1-x) \Delta_{d,j}(\text{O}^{2-}) \\ &+$$

When *M* is Co, the mathematical models for estimating the $\Delta H_{f,298}^{\theta}$, $\Delta G_{f,298}^{\theta}$, and C_{p} of lithium-rich layered $x \text{Li}_{2}\text{MnO}_{3} \cdot (1-x)\text{Li}MO_{2}$ materials using the group contribution method are shown in Eq.(8~10), respectively:

$$\Delta H^{\theta}_{,298}(x \text{Li}_2 \text{MnO}_3 \cdot (1-x) \text{LiCoO}_2) = -514.463x - 678.200$$
(8)

$$\Delta G^{\theta}_{f_{298}}(x \text{Li}_2 \text{MnO}_3 \cdot (1-x) \text{LiCoO}_2) = -508.336x - 615.100$$
(9)

$$C_{p}(xLi_{2}MnO_{3} \cdot (1-x)LiCoO_{2}) = 6.040x + 106.101 + (80.290x - 13.865) \times 10^{-3}T + (0.038x - 1.650) \times 10^{6}/T^{2} + (13.808x - 4.687) \times 10^{-6}T^{2}$$
(10)

When *M* is Fe, the mathematical models for estimating the $\Delta H^{\theta}_{f,298}$, $\Delta G^{\theta}_{f,298}$, and C_{p} of lithium-rich layered $x \text{Li}_{2}\text{MnO}_{3} \cdot (1-x)$ Li*M*O₂ materials using the group contribution method are shown in Eq.(11~13), respectively:

$$\Delta H^{\theta}_{f_{2}98}(x \operatorname{Li}_{2} \operatorname{MnO}_{3} \cdot (1-x) \operatorname{LiFeO}_{2}) = -429.598x - 763.065 \quad (11)$$

$$\Delta G^{\theta}_{f,298}(x \text{Li}_2 \text{MnO}_3 \cdot (1-x) \text{LiFeO}_2) = -404.872x - 718.564 \quad (12)$$

$$C_{p}(xLi_{2}MnO_{3}\cdot(1-x)LiFeO_{2})=23.580x+88.561 +(34.887x+31.538)\times10^{-3}T+(-0.193x-1.419)\times10^{6}/T^{2} +(4.192x+4.929)\times10^{-6}T^{2}$$
(13)

When *M* is Ni_{1/2}Mn_{1/2}, the mathematical models for estimating the $\Delta H^{\theta}_{f,298}$, $\Delta G^{\theta}_{f,298}$, and $C_{\rm p}$ of lithium-rich layered $x {\rm Li}_2 {\rm MnO}_3 \cdot (1-x) {\rm Li}M{\rm O}_2$ materials using the group contribution method are shown in Eq.(14~16), respectively:

$$\Delta H_{f,298}^{\theta}(x \text{Li}_2 \text{MnO}_3 \cdot (1-x) \text{LiNi}_{1/2} \text{Mn}_{1/2} \text{O}_2) = -463.816x - 728.847$$
(14)

$$\Delta G^{\circ}_{f,298}(xL1_2MnO_3 \cdot (1-x)L1N1_{1/2}Mn_{1/2}O_2) = -446.979x - 676.458$$
(15)

$$C_{p}(xLi_{2}MnO_{3}\cdot(1-x)LiNi_{1/2}Mn_{1/2}O_{2})=30.746x+81.395 +(30.527x+35.899)(10^{-3})T+(-0.422x-1.191)(10^{6})/T^{2} +(3.455x+5.666)(10^{-6})T^{2}$$
(16)

2.2 Mathematical models for estimating $\Delta H^{\theta}_{f,298}$, and $\Delta G^{\theta}_{f,298}$, and C_{p} of lithium-rich manganese-based xLi[Li_{1/3}Mn_{2/3}]O₂·(1-x)LiMO₂ solid solution materials

The chemical valences of Li, O, Mn, and *M* in the lithium-rich manganese-based solid solution materials of $xLi[Li_{1/3}Mn_{2/3}]O_2 \cdot (1-x)LiMO_2$ (0<x<1, *M*=Ni, Co, Cr, Fe, Al, Mg, Ni-Co, Ni-Mn, Ni-Co-Mn...) are +1, -2, +4, and +3, respectively. The most studied systems of the lithium-rich manganese-based $xLi[Li_{1/3}Mn_{2/3}]O_2 \cdot (1-x)LiMO_2$ material are Ni-Mn, Ni-Co-Mn. Common layered lithium-rich materials include $xLi[Li_{1/3}Mn_{2/3}]O_2 \cdot (1-x)LiNi_{1/3}Co_{1/3}Mn_{1/3}O_2$ and $xLi[Li_{1/3}Mn_{2/3}]O_2 \cdot (1-x)LiNi_{1/3}O_2$.

The mathematical models for estimating the $\Delta H_{1,298}^{\theta}$, $\Delta G_{1,298}^{\theta}$, and C_{p} of lithium-rich manganese-based xLi[Li_{1/3}Mn_{2/3}]O₂·(1-x)LiMO₂·(1-x)LiMO₂ (0<x<1, M=Ni, Co, Cr, Fe, Al, Mg, Ni-Co, Ni-Mn, Ni-Co-Mn...) solid solution materials by the group contribution method are shown in Eq.(17~19):

$$\Delta H^{\theta}_{f,298}(x \operatorname{Li}[\operatorname{Li}_{1/3}\operatorname{Mn}_{2/3}]O_{2} \cdot (1-x)\operatorname{Li}MO_{2}) = 4/3x \Delta H^{\theta}_{f,298}(\operatorname{Li}^{+}) + 2/3x \Delta H^{\theta}_{f,298}(\operatorname{Mn}^{4+}) + 2x \Delta H^{\theta}_{f,298}(\operatorname{O}^{2-}) + (1-x) \Delta H^{\theta}_{f,298}(\operatorname{Li}^{+}) + (1-x) \Delta H^{\theta}_{f,298}(M^{3+}) + 2(1-x) \Delta H^{\theta}_{f,298}(\operatorname{O}^{2-})$$
(17)

$$\Delta G^{\theta}_{f,298}(x \operatorname{Li}[\operatorname{Li}_{1/3}\operatorname{Mn}_{2/3}]O_{2} \cdot (1-x)\operatorname{Li}MO_{2}) = 4/3x \Delta G^{\theta}_{f,298}(\operatorname{Li}^{+}) + 2/3x \Delta G^{\theta}_{f,298}(\operatorname{Mn}^{4+}) + 2x \Delta G^{\theta}_{f,298}(\operatorname{O}^{2-}) + (1-x) \Delta G^{\theta}_{f,298}(\operatorname{Li}^{+}) + (1-x) \Delta G^{\theta}_{f,298}(\operatorname{M}^{3+}) + 2(1-x) \Delta G^{\theta}_{f,298}(\operatorname{O}^{2-})$$
(18)

$$C_{p}(xLi[Li_{1/3}Mn_{2/3}]O_{2} \cdot (1-x)LiMO_{2}) = \sum_{j}n_{j}\Delta_{a,j} + (\sum_{j}n_{j}\Delta_{b,j} \times 10^{-3})T + (\sum_{j}n_{j}\Delta_{c,j} \times 10^{6})/T^{2} + (\sum_{j}n_{j}\Delta_{d,j} \times 10^{-6})T^{2}$$

$$\sum_{j}n_{j}\Delta_{a,j} = 4/3x\Delta_{a,j}(Li^{+}) + 2/3x\Delta_{a,j}(Mn^{4+}) + 2x\Delta_{a,j}(O^{2-}) + (1-x)\Delta_{a,j}(Li^{+}) + (1-x)\Delta_{a,j}(M^{3+}) + 2(1-x)\Delta_{a,j}(O^{2-})$$

$$\sum_{j}n_{j}\Delta_{b,j} = 4/3x\Delta_{b,j}(Li^{+}) + 2/3x\Delta_{b,j}(Mn^{4+}) + 2x\Delta_{b,j}(O^{2-}) + (1-x)\Delta_{b,j}(Li^{+}) + (1-x)\Delta_{b,j}(M^{3+}) + 2(1-x)\Delta_{b,j}(O^{2-})$$

$$\sum_{j}n_{j}\Delta_{c,j} = 4/3x\Delta_{c,j}(Li^{+}) + 2/3x\Delta_{c,j}(Mn^{4+}) + 2x\Delta_{c,j}(O^{2-}) + (1-x)\Delta_{c,j}(Li^{+}) + (1-x)\Delta_{c,j}(M^{3+}) + 2(1-x)\Delta_{c,j}(O^{2-})$$

$$\sum_{j}n_{j}\Delta_{d,j} = 4/3x\Delta_{d,j}(Li^{+}) + 2/3x\Delta_{d,j}(Mn^{4+}) + 2x\Delta_{d,j}(O^{2-})$$

+ $(1-x)\Delta_{d,j}(Li^{+})$ + $(1-x)\Delta_{d,j}(M^{3+})$ + $2(1-x)\Delta_{d,j}(O^{2-})$ (19)

When *M* is $Ni_{1/2}Mn_{1/2}$, the mathematical models for estimating the $\Delta H^{\theta}_{f,298}$, $\Delta G^{\theta}_{f,298}$, and C_{p} of lithium-rich

manganese-based $x \text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]O_2 \cdot (1-x)\text{Li}MO_2$ solid solution materials by the group contribution method are shown in Eq.(20~22):

$$\Delta H^{\theta}_{f,298}(x \operatorname{Li}[\operatorname{Li}_{1/3} \operatorname{Mn}_{2/3}]O_2 \cdot (1-x) \operatorname{LiNi}_{1/2} \operatorname{Mn}_{1/2}O_2) = -66.262x - 728.847$$
(20)

$$\Delta G^{\theta}_{f_{2}98}(x \text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]\text{O}_{2} \cdot (1-x)\text{Li}\text{Ni}_{1/2}\text{Mn}_{1/2}\text{O}_{2}) = -72.500x - 676.458$$
(21)

$$C_{p}(xLi[Li_{1/3}Mn_{2/3}]O_{2} \cdot (1-x)LiNi_{1/2}Mn_{1/2}O_{2}) = -6.634x + 81.395 + (8.385x + 35.899)(10^{-3})T + (0.116x - 1.191)(10^{6})/T^{2} + (0.415x + 5.666)(10^{-6})T^{2}$$
(22)

When *M* is Ni_{1/3}Co_{1/3}Mn_{1/3}, the mathematical models for estimating the $\Delta H_{f_{298}}^{\theta}$, $\Delta G_{f_{298}}^{\theta}$, and C_{p} of lithium-rich manganesebased *x*Li[Li_{1/3}Mn_{2/3}]O₂ · (1 – *x*)Li*M*O₂ solid solution materials by the group contribution method are shown in Eq.(23~25):

$$\Delta H^{\theta}_{f,298}(x \operatorname{Li}[\operatorname{Li}_{1/3}\operatorname{Mn}_{2/3}]O_{2} \cdot (1-x)\operatorname{LiNi}_{1/3}\operatorname{Co}_{1/3}\operatorname{Mn}_{1/3}O_{2}) = -83.145x - 711.964$$
(23)
$$\Delta G^{\theta} \quad (x \operatorname{Li}[\operatorname{Li}_{1/3}\operatorname{Mn}_{1/3}]O_{2} \cdot (1-x)\operatorname{LiNi}_{1/3}Co_{1/3}\operatorname{Mn}_{1/3}O_{2}) = -83.145x - 711.964$$

$$\Delta G_{f_{298}}(x \text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]\text{O}_{2} \cdot (1-x)\text{Li}\text{Ni}_{1/3}\text{Co}_{1/3}\text{Min}_{1/3}\text{O}_{2})$$

$$= -92.952x - 656.005 \qquad (24)$$

$$C(x \text{Li}[\text{Li}, \text{Mn}, \text{IO}; (1-x)\text{Li}\text{Ni}, \text{Co}, \text{Mn}, \text{O})$$

$$\begin{aligned} & = -14.869x + 89.630 + (24.973x + 19.311)(10^3)T \\ & + (0.269x - 1.344)(10^6)/T^2 + (3.866x + 2.215)(10^6)T^2 \end{aligned} (25)$$

2.3 Mathematical models for estimating $\Delta H^{\theta}_{f,298}$, $\Delta G^{\theta}_{f,298}$, and C_{p} of the layered spinel composite solid solution material

The chemical valences of Li, O in layered spinel $x\text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]\text{O}_2$ · $(1 - x)\text{Li}\text{Mn}_2\text{O}_4$ (0<x<1) composite solid solution material are +1, -2, respectively. The Mn element present in Li[Li_{1/3}Mn_{2/3}]O₂ has a valence of +4, and LiMn₂O₄ has a Mn³⁺ and Mn⁴⁺ content of 50%.

The mathematical models for estimating the $\Delta H_{f,298}^{\theta}$, $\Delta G_{f,298}^{\theta}$, and $C_{\rm p}$ of the layered spinel $x {\rm Li}[{\rm Li}_{1/3}{\rm Mn}_{2/3}]{\rm O}_2 \cdot (1-x){\rm Li}{\rm Mn}_2{\rm O}_4$ composite solid solution materials using the group contribution method are shown in Eq.(26~28), respectively:

$$\begin{split} \Delta H^{\theta}_{f,298}(x \text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]\text{O}_{2} \cdot (1-x) \text{Li}\text{Mn}_{2}\text{O}_{4}) = 4/3x \Delta H^{\theta}_{f,298}(\text{Li}^{+}) \\ + 2/3x \Delta H^{\theta}_{f,298}(\text{Mn}^{4+}) + 2x \Delta H^{\theta}_{f,298}(\text{O}^{2-}) + (1-x) \Delta H^{\theta}_{f,298}(\text{Li}^{+}) \\ + 2\times 0.5(1-x) \Delta H^{\theta}_{f,298}(\text{Mn}^{3+}) + 2\times 0.5(1-x) \Delta H^{\theta}_{f,298}(\text{Mn}^{4+}) \\ + 4(1-x) \Delta H^{\theta}_{f,298}(\text{O}^{2-}) & (26) \\ \Delta G^{\theta}_{f,298}(x \text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]\text{O}_{2} \cdot (1-x) \text{Li}\text{Mn}_{2}\text{O}_{4}) \\ = 4/3x \Delta G^{\theta}_{f,298}(\text{O}^{2-}) + (1-x) \Delta G^{\theta}_{f,298}(\text{Mn}^{4+}) \\ + 2x \Delta G^{\theta}_{f,298}(\text{O}^{2-}) + (1-x) \Delta G^{\theta}_{f,298}(\text{Li}^{+}) + \\ 2\times 0.5(1-x) \Delta G^{\theta}_{f,298}(\text{Mn}^{3+}) + 2\times 0.5(1-x) \Delta G^{\theta}_{f,298}(\text{Mn}^{4+}) \\ + 4(1-x) \Delta G^{\theta}_{f,298}(\text{O}^{2-}) & (27) \\ C_{p}(x \text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]\text{O}_{2} \cdot (1-x) \text{Li}\text{Mn}_{2}\text{O}_{4}) \\ = \sum_{j} n_{j} \Delta_{a,j} + (\sum_{j} n_{j} \Delta_{b,j} \times 10^{-3} T + (\sum_{j} n_{j} \Delta_{c,j} \times 10^{6})/T^{2} \\ + (\sum_{j} n_{j} \Delta_{a,j}(\text{Li}^{+}) + 2/3x \Delta_{a,j}(\text{Mn}^{4+}) + 2x \Delta_{a,j}(\text{O}^{2-}) \\ + (1-x) \Delta_{a,j}(\text{Li}^{+}) + 2\times 0.5(1-x) \Delta_{a,j}(\text{Mn}^{3+}) \\ + 2\times 0.5(1-x) \Delta_{a,j}(\text{Mn}^{4+}) + 4(1-x) \Delta_{a,j}(\text{O}^{2-}) \\ \sum_{j} n_{j} \Delta_{b,j} = 4/3x \Delta_{b,j}(\text{Li}^{+}) + 2/3x \Delta_{b,j}(\text{Mn}^{3+}) \\ + 2\times 0.5(1-x) \Delta_{b,j}(\text{Mn}^{4+}) + 4(1-x) \Delta_{b,j}(\text{O}^{2-}) \\ \sum_{j} n_{j} \Delta_{c,j} = 4/3x \Delta_{c,j}(\text{Li}^{+}) + 2/3x \Delta_{c,j}(\text{Mn}^{3+}) \\ + 2\times 0.5(1-x) \Delta_{b,j}(\text{Mn}^{4+}) + 4(1-x) \Delta_{b,j}(\text{O}^{2-}) \\ \sum_{j} n_{j} \Delta_{c,j} = 4/3x \Delta_{c,j}(\text{Li}^{+}) + 2/3x \Delta_{c,j}(\text{Mn}^{3+}) \\ + 2\times 0.5(1-x) \Delta_{b,j}(\text{Mn}^{4+}) + 4(1-x) \Delta_{b,j}(\text{O}^{2-}) \\ \sum_{j} n_{j} \Delta_{c,j} = 4/3x \Delta_{c,j}(\text{Li}^{+}) + 2/3x \Delta_{c,j}(\text{Mn}^{3+}) \\ + 2\times 0.5(1-x) \Delta_{b,j}(\text{Mn}^{4+}) + 4(1-x) \Delta_{b,j}(\text{O}^{2-}) \\ \sum_{j} n_{j} \Delta_{c,j} = 4/3x \Delta_{c,j}(\text{Li}^{+}) + 2/3x \Delta_{c,j}(\text{Mn}^{3+}) \\ + 2\times 0.5(1-x) \Delta_{c,j}(\text{Mn}^{3+}) \\ + 2\times 0.5(1$$

$$\sum_{j} n_{j} \Delta_{d,j} = 4/3x \Delta_{d,j} (\mathrm{Li}^{+}) + 2/3x \Delta_{d,j} (\mathrm{Mn}^{4+}) + 2x \Delta_{d,j} (\mathrm{O}^{2-}) + (1-x) \Delta_{d,j} (\mathrm{Li}^{+}) + 2 \times 0.5(1-x) \Delta_{d,j} (\mathrm{Mn}^{3+}) + 2 \times 0.5(1-x) \Delta_{d,j} (\mathrm{Mn}^{4+}) + 4(1-x) \Delta_{d,j} (\mathrm{O}^{2-})$$
(28)

In Eq. (28), the values of the group contribution can be considered as the contributions in the layered spinel $x \text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]\text{O}_2$ · $(1 - x)\text{Li}\text{Mn}_2\text{O}_4$ composite solid solution material. The mathematical models of $\Delta H^{\theta}_{f,298}$, $\Delta G^{\theta}_{f,298}$, and C_p are shown in Eq.(29~31), respectively:

$$\Delta H^{\theta}_{f,298}(x \text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]\text{O}_{2} \cdot (1-x)\text{Li}\text{Mn}_{2}\text{O}_{4}) = 446.175x - 1241.284$$
(29)
$$\Delta G^{\theta}_{f,299}(x \text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]\text{O}_{2} \cdot (1-x)\text{Li}\text{Mn}_{2}\text{O}_{4})$$

$$=454.248x-1203.205$$
 (30)
C (xLi[Li, Mn_2]O_2:(1-x)LiMn_Q.)

$$= -58.806x + 133.567 + (-33.131x + 77.414)(10^{-3})T + (0.823x - 1.898)(10^6)/T^2 + (3.508x + 2.573)(10^6)/T^2$$
(31)

In the layered spinel $x \text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]\text{O}_2 \cdot (1-x)\text{Li}\text{Ni}_{0.5}\text{Mn}_{1.5}\text{O}_4$ (0<x<1) composite solid solution material, the chemical valences of Li and O are +1 and -2, respectively. The mathematical models for estimating the $\Delta H^{\theta}_{f,298}$, $\Delta G^{\theta}_{f,298}$, and C_p of the layered spinel $x \text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]\text{O}_2 \cdot (1-x)\text{Li}\text{Ni}_{0.5}\text{Mn}_{1.5}\text{O}_4$ composite solid solution materials using the group contribution method are shown in Eq.(32~34), respectively:

$$\Delta H^{\theta}_{f,298}(x \text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]\text{O}_{2} \cdot (1-x) \text{LiNi}_{0.5}\text{Mn}_{1.5}\text{O}_{4}) \\ = 4/3x \Delta H^{\theta}_{f,298}(\text{Li}^{+}) + 2/3x \Delta H^{\theta}_{f,298}(\text{Mn}^{4+}) \\ + 2x \Delta H^{\theta}_{f,298}(\text{O}^{2-}) + (1-x) \Delta H^{\theta}_{f,298}(\text{Li}^{+}) \\ + 0.5(1-x) \Delta H^{\theta}_{f,298}(\text{Ni}^{2+}) + 1.5(1-x) \Delta H^{\theta}_{f,298}(\text{Mn}^{4+}) \\ 4(1-x) \Delta H^{\theta}_{f,298}(\text{O}^{2-})$$
(32)
$$\Delta G^{\theta}_{f,298}(x \text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]\text{O}_{2} \cdot (1-x) \text{LiNi}_{0.5}\text{Mn}_{1.5}\text{O}_{4}) \\ = 4/3x \Delta G^{\theta}_{f,298}(\text{Li}^{+}) + 2/3x \Delta G^{\theta}_{f,298}(\text{Mn}^{4+}) \\ + 2x \Delta G^{\theta}_{f,298}(\text{O}^{2-}) + (1-x) \Delta G^{\theta}_{f,298}(\text{Li}^{+}) \\ + 0.5(1-x) \Delta G^{\theta}_{f,298}(\text{Ni}^{2+}) + 1.5(1-x) \Delta G^{\theta}_{f,298}(\text{Mn}^{4+})$$

$$+4 (1-x) \Delta G^{\theta}_{f,298}(O^{2-})$$

$$C_{p}(x \text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]O_{2} \cdot (1-x) \text{Li}\text{Ni}_{0.5}\text{Mn}_{1.5}O_{4})$$

$$= \sum n \Delta + (\sum n \Delta \times 10^{-3})T + (\sum n \Delta \times 10^{6})/T^{2}$$
(33)

$$= \sum_{j} r_{j} - r_{a,j} + \sum_{j} r_{j} - r_{b,j} + r_{$$

$$\sum_{j} n_{j} \Delta_{a_{j}} - 4/3\lambda \Delta_{a_{j}}(L1) + 2/3\lambda \Delta_{a_{j}}(L11) + 2/3\lambda \Delta_{a_{j}}(L11)$$

$$\sum_{j} h_{\Delta b_{j}}^{-2} - 4/3 \chi \Delta_{b_{j}}^{-1} (L^{+}) + 2/3 \chi \Delta_{b_{j}}^{-1} (N^{+}) + 2\chi \Delta_{b_{j}}^{-1} (O^{-}) + (1-\chi) \Delta_{b_{j}}^{-1} (L^{+}) + 0.5(1-\chi) \Delta_{b_{j}}^{-1} (N^{+}) + 1.5(1-\chi) \Delta_{b_{j}}^{-1} (Mn^{4+}) + 4(1-\chi) \Delta_{b_{j}}^{-1} (O^{-})$$

$$\sum_{j} n \Delta_{j}^{-2} - 4/3 \chi \Delta_{j}^{-1} (L^{+}) + 2/3 \chi \Delta_{j}^{-1} (Mn^{4+}) + 2\chi \Delta_{j}^{-1} (O^{-})$$

$$\sum_{j} n_{j} \Delta_{cj}(\mathrm{Li}^{+}) + 0.5(1-x)\Delta_{cj}(\mathrm{Li}^{+}) + 0.5(1-x)\Delta_{cj}(\mathrm{Ni}^{2+}) \\ + 1.5(1-x)\Delta_{cj}(\mathrm{Mn}^{4+}) + 4(1-x)\Delta_{cj}(\mathrm{O}^{2-}) \\ \sum_{j} n_{j}\Delta_{d,j} = 4/3x\Delta_{d,j}(\mathrm{Li}^{+}) + 2/3x\Delta_{d,j}(\mathrm{Mn}^{4+}) + 2x\Delta_{d,j}(\mathrm{O}^{2-}) \\ + (1-x)\Delta_{d,j}(\mathrm{Li}^{+}) + 0.5(1-x)\Delta_{d,j}(\mathrm{Ni}^{2+}) \\ + 1.5(1-x)\Delta_{d,j}(\mathrm{Mn}^{4+}) + 4(1-x)\Delta_{d,j}(\mathrm{O}^{2-})$$

In Eq.(34), the values of the group contribution can be obtained as the layered spinel xLi[Li_{1/3}Mn_{2/3}]O₂(1-x)LiNi_{0.5}Mn_{1.5}O₄ composite solid solution material. The mathematical models of $\Delta H_{f,298}^{\theta}, \Delta G_{f,298}^{\theta}$, and C_{p} are shown in Eq.(35~37), respectively:

$$\Delta H^{\theta}_{f_{298}}(x \text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]\text{O}_{2} \cdot (1-x) \text{LiNi}_{0.5}\text{Mn}_{1.5}\text{O}_{4}) = 366.851x - 1161.960$$
(35)
$$\Delta G^{\theta}_{f_{298}}(x \text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]\text{O}_{2} \cdot (1-x) \text{LiNi}_{0.5}\text{Mn}_{1.5}\text{O}_{4})$$

$C_{p}(xLi[Li_{1/3}Mn_{2/3}]O_{2} \cdot (1-x)LiNi_{0.5}Mn_{1.5}O_{4}) = -59.345x$	+
$134.106 + (-37.749x + 82.033)(10^{-3})T + (0.967x - 2.033)(10^{-3})T + (0.977x - 2.033)(10^{-3})T + ($	042)
$(10^6)/T^2 + (3.217x + 2.864)(10^{-6})T^2$	(37)

The $\Delta H^{\theta}_{f,298}$, $\Delta G^{\theta}_{f,298}$, and $C_{\rm p}$ of the common 63 lithium-rich materials were estimated by the mathematical models, as show in Table 7.

Sample	ΛH^{θ} /kI·mol ⁻¹	ΔG^{θ} /kI·mol ⁻¹	$C_{p,298}$ predicted by the group	$C_{p,298}$ predicted
Sample	Δ11 _{f,298} / KJ 11101	40 _{f,298} /KJ 11101	J·mol ⁻¹ ·K ⁻¹	$J \cdot mol^{-1} \cdot K^{-1}$
0.1Li,MnO ₃ ·0.9LiCoO,	-729.65	-665.93	86.13	79.70
$0.2Li_2MnO_3 \cdot 0.8LiCoO_2$	-781.09	-716.77	89.30	83.60
0.3Li ₂ MnO ₃ ·0.7LiCoO ₂	-832.54	-767.60	92.46	87.51
0.4Li ₂ MnO ₃ ·0.6LiCoO ₂	-883.99	-818.43	95.62	91.41
0.5Li ₂ MnO ₃ ·0.5LiCoO ₂	-935.43	-869.27	98.78	95.31
0.6Li ₂ MnO ₃ ·0.4LiCoO ₂	-986.88	-920.10	101.95	99.21
$0.7 \text{Li}_2 \text{MnO}_3 \cdot 0.3 \text{LiCoO}_2$	-1038.32	-970.94	105.11	103.11
$0.8 \text{Li}_2 \text{MnO}_3 \cdot 0.2 \text{LiCoO}_2$	-1089.77	-1021.77	108.27	107.02
$0.9 \text{Li}_2 \text{MnO}_3 \cdot 0.1 \text{LiCoO}_2$	-1141.22	-1072.60	111.43	110.92
0.1Li ₂ MnO ₃ ·0.9LiFeO ₂	-806.02	-759.05	85.64	82.74
$0.2 \text{Li}_2 \text{MnO}_3 \cdot 0.8 \text{LiFeO}_2$	-848.98	-799.54	88.85	86.30
$0.3 \text{Li}_2 \text{MnO}_3 \cdot 0.7 \text{LiFeO}_2$	-891.94	-840.03	92.07	89.87
$0.4 \text{Li}_2 \text{MnO}_3 \cdot 0.6 \text{LiFeO}_2$	-934.90	-880.51	95.29	93.43
$0.5 \text{Li}_2 \text{MnO}_3 \cdot 0.5 \text{LiFeO}_2$	-977.86	-921.00	98.51	97.00
$0.6 \text{Li}_2 \text{MnO}_3 \cdot 0.4 \text{LiFeO}_2$	-1020.82	-961.49	101.72	100.56
$0.7 \text{Li}_2 \text{MnO}_3 \cdot 0.3 \text{LiFeO}_2$	-1063.78	-1001.97	104.94	104.13
$0.8 \text{Li}_2 \text{MnO}_3 \cdot 0.2 \text{LiFeO}_2$	-1106.74	-1042.46	108.16	107.69
$0.9 \text{Li}_2 \text{MnO}_3 \cdot 0.1 \text{LiFeO}_2$	-1149.70	-1082.95	111.38	111.26
$0.1Li_2MnO_3$: $0.9LiNi_{1/2}Mn_{1/2}O_2$	-775.23	-721.16	82.73	80.65
$0.2Li_2MnO_3$: $0.8LiNi_{1/2}Mn_{1/2}O_2$	-821.61	-765.85	86.27	84.44
$0.3Li_2MnO_3 \cdot 0.7LiNi_{1/2}Mn_{1/2}O_2$	-867.99	-810.55	89.80	88.24
$0.4Li_2MnO_3 \cdot 0.6LiNi_{1/2}Mn_{1/2}O_2$	-914.37	-855.25	93.34	92.04
$0.5Li_2MnO_3 \cdot 0.5LiNi_{1/2}Mn_{1/2}O_2$	-960.76	-899.95	96.88	95.84
$0.6Li_2MnO_3 \cdot 0.4LiNi_{1/2}Mn_{1/2}O_2$	-1007.14	-944.65	100.42	99.63
$0.7Li_2MnO_3 \cdot 0.3LiNi_{1/2}Mn_{1/2}O_2$	-1053.52	-989.34	103.96	103.43
$0.8 Li_2 MnO_3 {\cdot} 0.2 LiNi_{1/2} Mn_{1/2}O_2$	-1099.90	-1034.04	107.50	107.23
$0.9 Li_2 MnO_3 {\cdot} 0.1 LiNi_{1/2} Mn_{1/2}O_2$	-1146.28	-1078.74	111.04	111.02
$0.1 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.9 Li Ni_{1/2} Mn_{1/2} O_2$	-735.47	-683.71	78.91	76.82
$0.2 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.8 Li Ni_{1/2} Mn_{1/2} O_2$	-742.10	-690.96	78.63	76.79
$0.3 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.7 Li Ni_{1/2} Mn_{1/2} O_2$	-748.73	-698.21	78.35	76.76
$0.4 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.6 Li Ni_{1/2} Mn_{1/2} O_2$	-755.35	-705.46	78.07	76.73
$0.5 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.5 Li Ni_{1/2} Mn_{1/2} O_2$	-761.98	-712.71	77.79	76.70
$0.6 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.4 Li Ni_{1/2} Mn_{1/2} O_2$	-768.60	-719.96	77.51	76.67
$0.7 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.3 Li Ni_{1/2} Mn_{1/2} O_2$	-775.23	-727.21	77.23	76.64
$0.8 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.2 Li Ni_{1/2} Mn_{1/2} O_2$	-781.86	-734.46	76.95	76.61
$0.9 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.1 Li Ni_{1/2} Mn_{1/2} O_2$	-788.48	-741.71	76.67	76.58
$0.1 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.9 Li Ni_{1/3} Co_{1/3} Mn_{1/3} O_2$	-720.28	-665.30	80.04	76.50
$0.2 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.8 Li Ni_{1/3} Co_{1/3} Mn_{1/3} O_2$	-728.59	-674.60	79.64	76.49
$0.3 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.7 Li Ni_{1/3} Co_{1/3} Mn_{1/3} O_2$	-736.91	-683.89	79.23	76.49
$0.4 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.6 Li Ni_{1/3} Co_{1/3} Mn_{1/3} O_2$	-745.22	-693.19	78.83	76.48
$0.5 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.5 Li Ni_{1/3} Co_{1/3} Mn_{1/3} O_2$	-753.54	-702.48	78.42	76.48
$0.6 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.4 Li Ni_{1/3} Co_{1/3} Mn_{1/3} O_2$	-761.85	-711.78	78.01	76.47

Table 7	Estimation results of ΔH^{θ} .	ΔG^{θ} and C	for lithium-rich materials
I HOIC /	Estimation results of Ent fr	$_{08}^{9}{f^{2}08}^{08}{f^{2}08}^{08}{n^{2}0}^{08}$	g for intinuum riten inatterians

Continued Table 7							
Sample	$\Delta H^{\theta}_{\rm f,298}/{\rm kJ}\cdot{\rm mol}^{-1}$	$\Delta G^{\theta}_{f,298}/\mathrm{kJ}\cdot\mathrm{mol}^{-1}$	$C_{p,298}$ predicted by the group contribution method/ J·mol ⁻¹ ·K ⁻¹	C _{p,298} predicted by the Kopp's rule∕ J⋅mol ⁻¹ ⋅K ⁻¹			
$0.7 \text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]\text{O}_2 \cdot 0.3 \text{Li}\text{Ni}_{1/3}\text{Co}_{1/3}\text{Mn}_{1/3}\text{O}_2$	-770.17	-721.07	77.61	76.47			
$0.8 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.2 Li Ni_{1/3} Co_{1/3} Mn_{1/3} O_2$	-778.48	-730.37	77.20	76.46			
$0.9 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.1 Li Ni_{1/3} Co_{1/3} Mn_{1/3} O_2$	-786.79	-739.66	76.80	76.46			
$0.1 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.9 Li Mn_2 O_4$	-1196.67	-1157.78	129.58	127.40			
$0.2 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.8 Li Mn_2 O_4$	-1152.05	-1112.36	123.67	121.75			
$0.3 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.7 Li Mn_2 O_4$	-1107.43	-1066.93	117.76	116.10			
$0.4 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.6 Li Mn_2 O_4$	-1062.81	-1021.51	111.85	110.45			
$0.5 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.5 Li Mn_2 O_4$	-1018.20	-976.08	105.94	104.80			
$0.6 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.4 Li Mn_2 O_4$	-973.58	-930.66	100.03	99.15			
$0.7 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.3 Li Mn_2 O_4$	-928.96	-885.23	94.12	93.50			
$0.8 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.2 Li Mn_2 O_4$	-884.34	-839.81	88.21	87.85			
$0.9 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.1 Li Mn_2 O_4$	-839.73	-794.38	82.30	82.20			
$0.1Li[Li_{1/3}Mn_{2/3}]O_2 \cdot 0.9LiNi_{0.5}Mn_{1.5}O_4$	-1125.27	-1037.49	129.87	126.23			
$0.2 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.8 Li Ni_{0.5} Mn_{1.5} O_4$	-1088.59	-1005.43	123.93	120.71			
$0.3Li[Li_{1/3}Mn_{2/3}]O_2 \cdot 0.7LiNi_{0.5}Mn_{1.5}O_4$	-1051.90	-973.37	117.99	115.19			
$0.4 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.6 Li Ni_{0.5} Mn_{1.5} O_4$	-1015.22	-941.31	112.04	109.67			
$0.5 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.5 Li Ni_{0.5} Mn_{1.5} O_4$	-978.53	-909.25	106.10	104.15			
$0.6 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.4 Li Ni_{0.5} Mn_{1.5} O_4$	-941.85	-877.19	100.16	98.63			
$0.7 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.3 Li Ni_{0.5} Mn_{1.5} O_4$	-905.16	-845.13	94.22	93.11			
$0.8 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.2 Li Ni_{0.5} Mn_{1.5} O_4$	-868.48	-813.08	88.28	87.59			
$0.9 Li [Li_{1/3} Mn_{2/3}] O_2 \cdot 0.1 Li Ni_{0.5} Mn_{1.5} O_4$	-831.79	-781.02	82.34	82.07			

3 Conclusions

1) The ΔH_{1298}^{θ} , ΔG_{1298}^{θ} , and $C_{p,298}$ of the three types of lithium-rich cathode materials consist of the thermodynamic group contribution values of Li⁺, Ni²⁺, Fe³⁺, Co²⁺, Co³⁺, Mn³⁺, and Mn⁴⁺. Three main types of lithium-rich cathode materials are explored: (1) lithium-rich layered $xLi_2MnO_3 \cdot (1-x)LiMO_2$ (0<x<1, *M*=Ni, Co, Cr, Fe, Al, Mg, Ni-Co, Ni-Mn, Ni-Co-Mn...) cathode material, (2) lithium-rich manganese-based $xLi_{1/3}MnO_{2/3} \cdot (1-x)LiMO_2$ (0<x<1, *M*=Ni, Co, Cr, Fe, Al, Mg, Ni-Co, Cr, Fe, Al, Mg, Ni-Co, Ni-Mn, Ni-Co-Mn...) solid solution material, and (3) layered spinel $xLi[Li_{1/3}Mn_{2/3}]O_2 \cdot (1-x)LiMn_2O_4$ (0<x<1) composite solid solution material and layered spinel $xLi[Li_{1/3}Mn_{2/3}]O_2 \cdot (1-x)LiNi_{0.5}Mn_{1.5}O_4$ (0<x<1) composite solid solution material.

2) The mathematical models for estimating the $\Delta H^{\theta}_{f,298}$, $\Delta G^{\theta}_{f,298}$, and $C_{\rm p}$ of three types of lithium-rich cathode materials are constructed for the first time by the group contribution method.

3) Based on the satisfactory results, the $\Delta H^{\theta}_{f,298}$, $\Delta G^{\theta}_{f,298}$, and $C_{\rm p}$ of three types of lithium-rich ${\rm Li}_{1+x}M_{1-x}O_2$ cathode materials are calculated, and those of 63 common lithium-rich ${\rm Li}_{1+x}M_{1-x}O_2$ materials can also be estimated by the mathematical models.

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基于基团贡献法估算(富锂)锂离子电池正极材料Li_{1+x}M_{1+x}O₂的热力学性质

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摘 要: 富锂Li_{1+x} $M_{1,x}$ O₂材料的研究主要集中在其结构和电化学性能上,而很少关注其热力学性能。开发具有高能量密度和容量的新型 富锂材料取决于这种材料的结构、热力学性质和电化学性质之间的固有关系。对富锂材料Li_{1+x} $M_{1,x}$ O₂的热力学性质了解不足,使得新型 Li_{1+x} $M_{1,x}$ O₂材料的开发和利用受到限制。鉴于Li_{1+x} $M_{1,x}$ O₂材料缺乏热力学数据,根据基团贡献方法的原理对LiAlO₂进行拆分。基于热力 学原理,提出了用于估计LiAlO₂的 ΔG^{θ}_{f298} 、 ΔH^{θ}_{f298} 和 C_{p} 的数学模型。采用基团贡献法估算了56种固体无机化合物的 ΔG^{θ}_{f298} 和 ΔH^{θ}_{f298} 以 及54种固体无机化合物的 $C_{p,298}$,以检验该模型的可靠性和适用性。利用基团贡献法估算了固体无机化合物的数学模型。利用基团贡献 法拟合的基团参数选择的实验数据准确可靠。在结果令人满意的基础上,建立了用于估算3种类型的Li_{1+x} $M_{1,x}$ O₂材料的 ΔG^{θ}_{f298} , ΔH^{θ}_{f298} 和 $C_{p,298}$ 。

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